



# What Numerical Modelling Tells Us About Dislocation Creep in Earth Mantle

Philippe Carrez

Francesca Boioli, Patrick Cordier, Karine Gouriet, Pierre Hirel, Antoine Kraych, Sebastian Ritterbex

Unité Matériaux et Transformations, UMR 8207 CNRS/Université  
Villeneuve d'Ascq, France

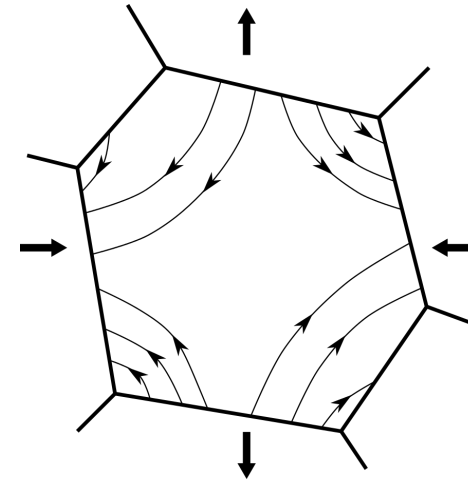


# Various creep mechanisms (dependant on temperature, stress, ...)

## Diffusion creep

Bulk diffusion (Nabarro-Herring creep)

Grain boundary diffusion (Coble creep)



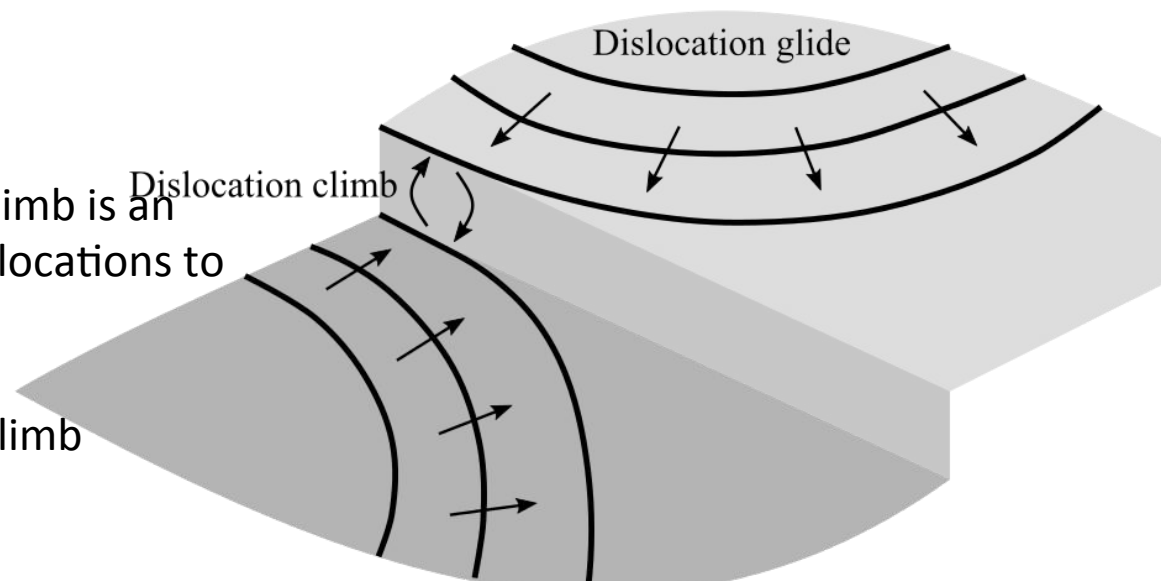
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## Dislocation creep

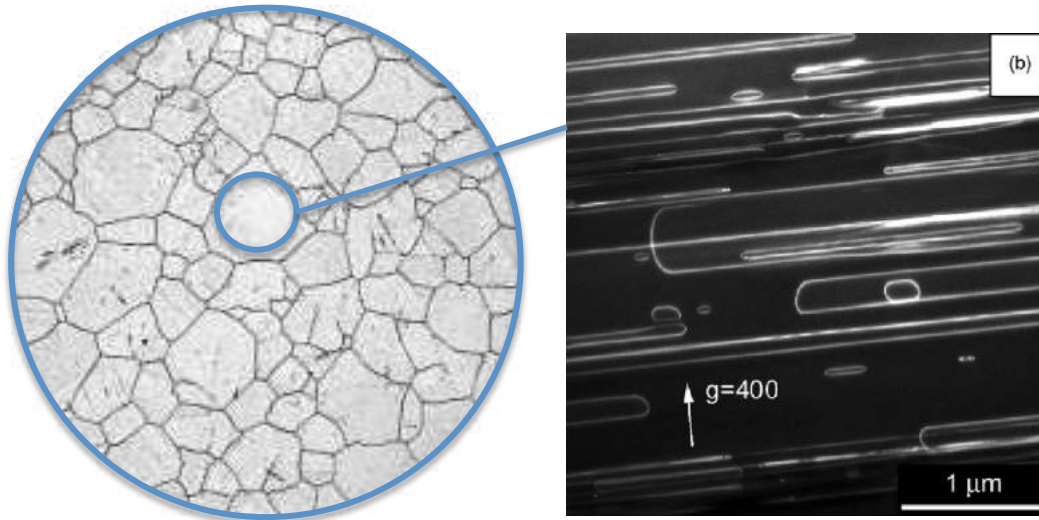
dislocation glide

Climb-assisted glide — here the climb is an enabling mechanism, allowing dislocations to get around obstacles (Weertman)

Climb — strain accomplished by climb

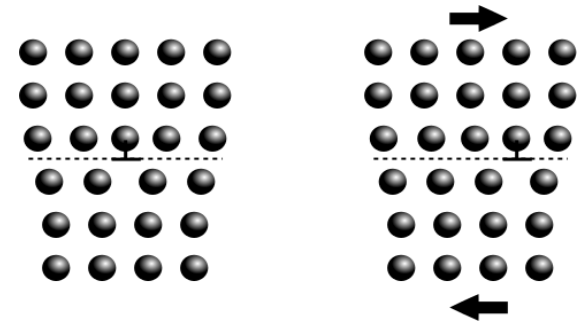


# A few words about dislocation

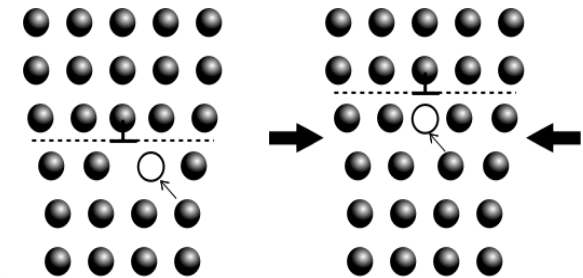


Olivine (P. Raterron)

**Glide**

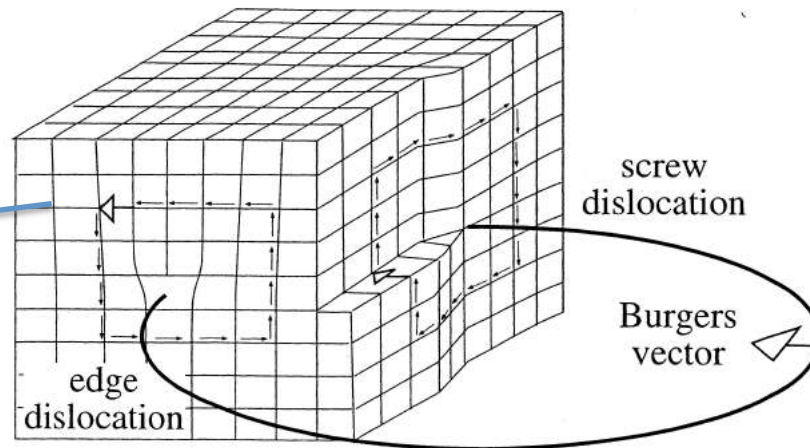
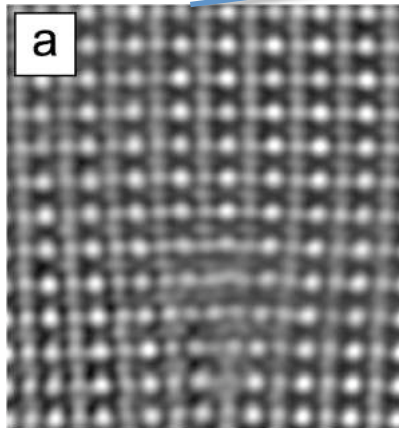


**Climb**



Jia et al. 2005

**a**

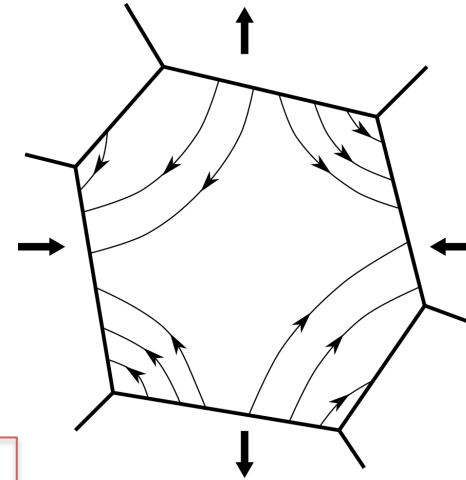


# Various creep mechanisms (dependant on temperature, stress, ...)

## Diffusion creep

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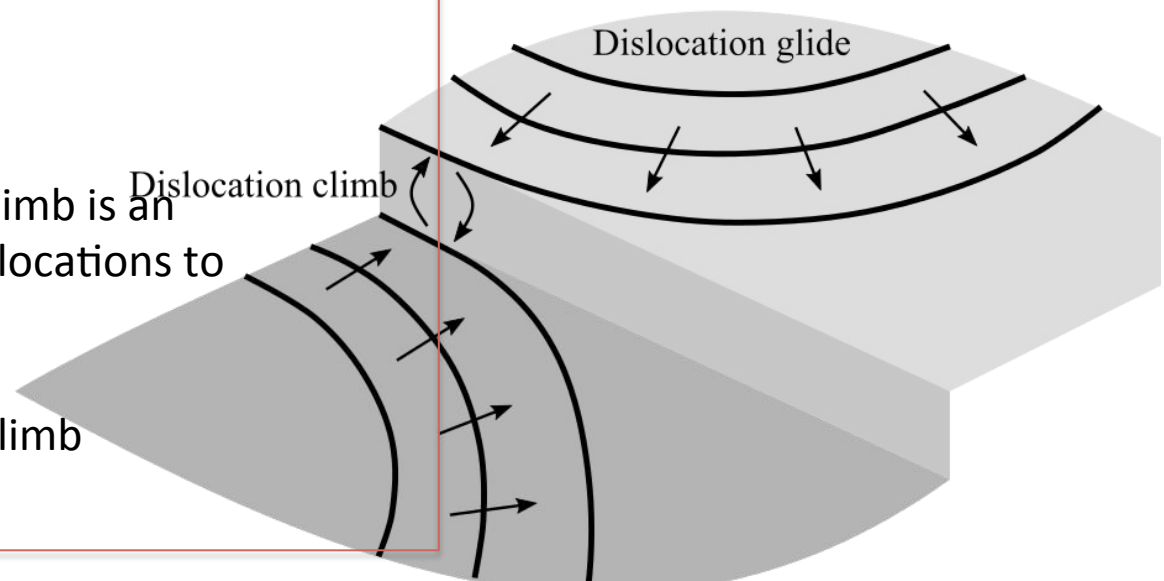


## Dislocation creep

dislocation glide

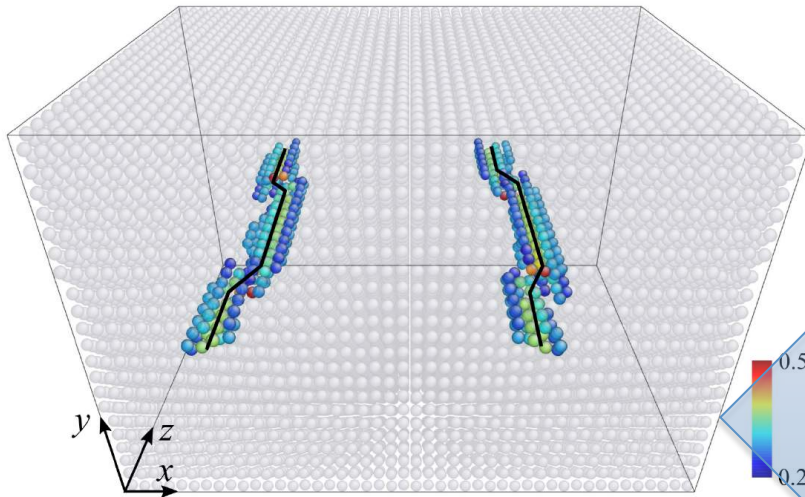
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Climb — strain accomplished by climb



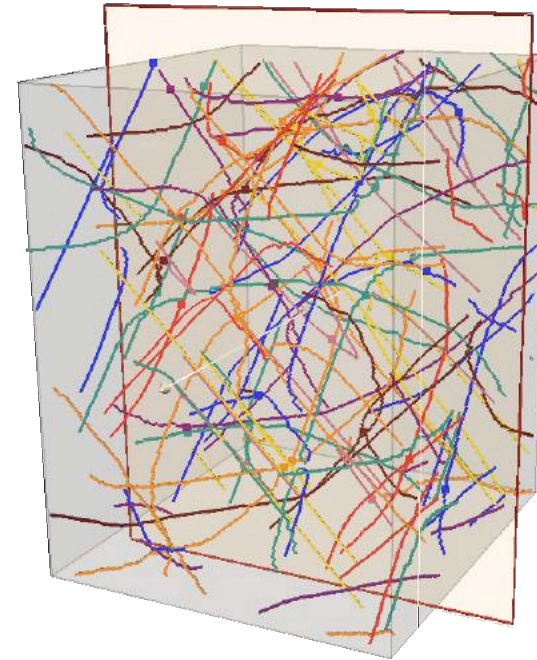
# Multi-Scale model of dislocation creep

Atomistic calculations



- dislocation core properties (effect of high Pressure)
- Dislocation mobility (effect of temperature on glide velocity)

Dislocation Dynamic (DD) simulation



Mesoscale simulation (grain scale)  
-> Collective behavior

Introduction of Climb (Steady state conditions,  
Constant vacancy concentration)

-> Rheological law (strain rate vs stress)

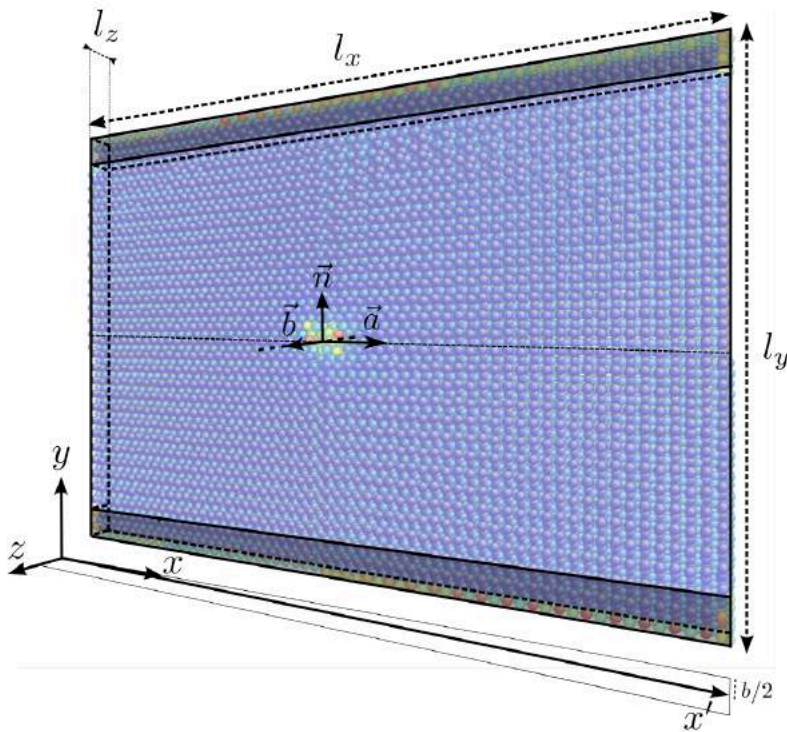


# Computational details for atomic scale calculations

Atomistic calculations performed with LAMMPS using a classical pairwise potential

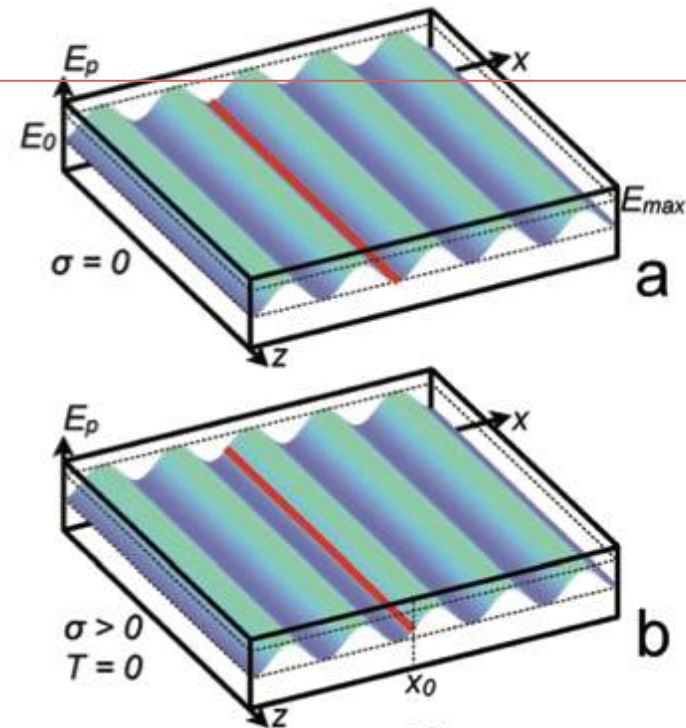
$$U_{ij}(R_{ij}) = \frac{z_i z_j}{R_{ij}} + b_{ij} \exp\left(-\frac{R_{ij}}{\rho_{ij}}\right) - \frac{c_{ij}}{R_{ij}^6}$$

$R_{ij}$  interatomic distance and  $z_i$  charge

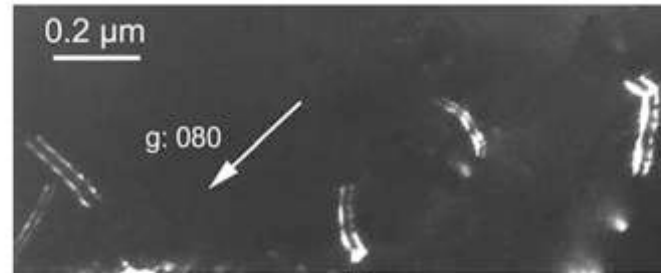
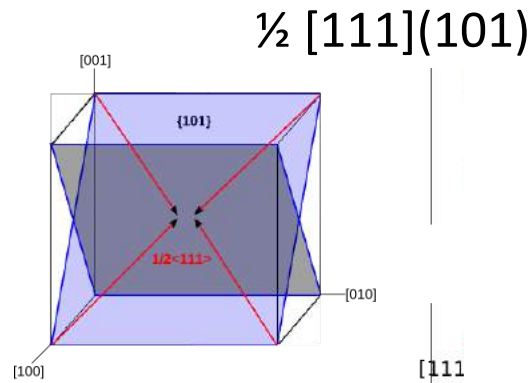


Atomic arrangement of atoms in the vicinity of the defect => Core structure

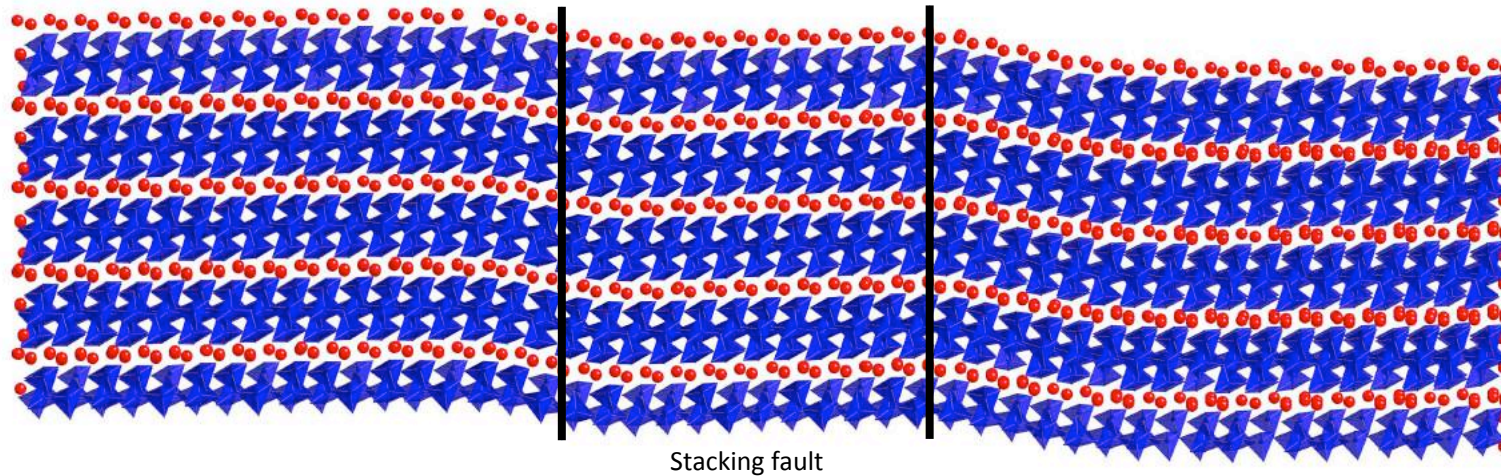
Peierls potential  $V_p$  (eV/Å) and associated Peierls stress



# Dislocation in wadsleyite (15 GPa)



Courtesy E. Thurel





# Dislocation core structure and Peierls stress



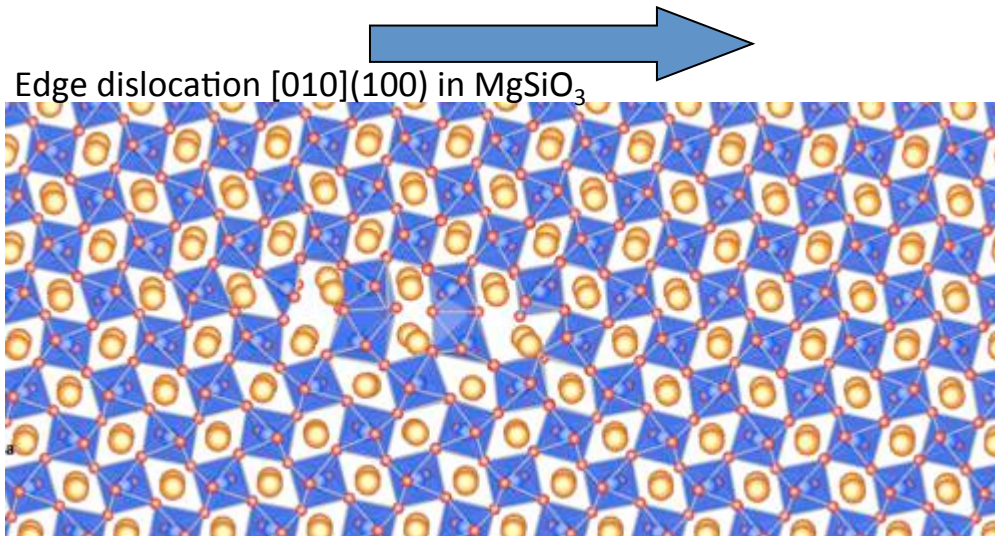
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Acta Materialia 79 (2014) 117–125



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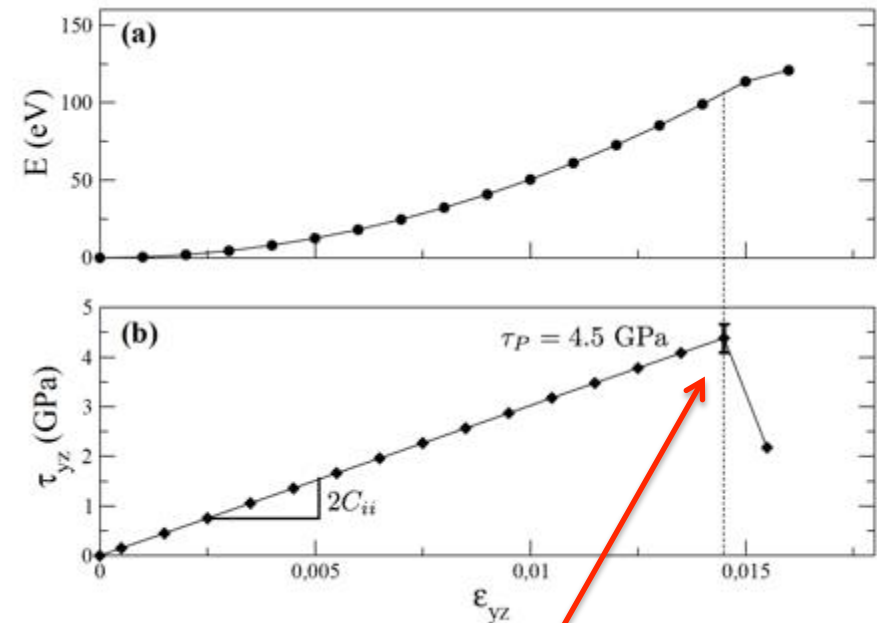


Atomic core structure and mobility of [100](010) and [010](100) dislocations in  $\text{MgSiO}_3$  perovskite

P. Hirel\*, A. Kraych, P. Carrez, P. Cordier

Unité Matériaux et Transformations, Bât. C6, Université de Lille 1, 59655 Villeneuve d'Ascq, France

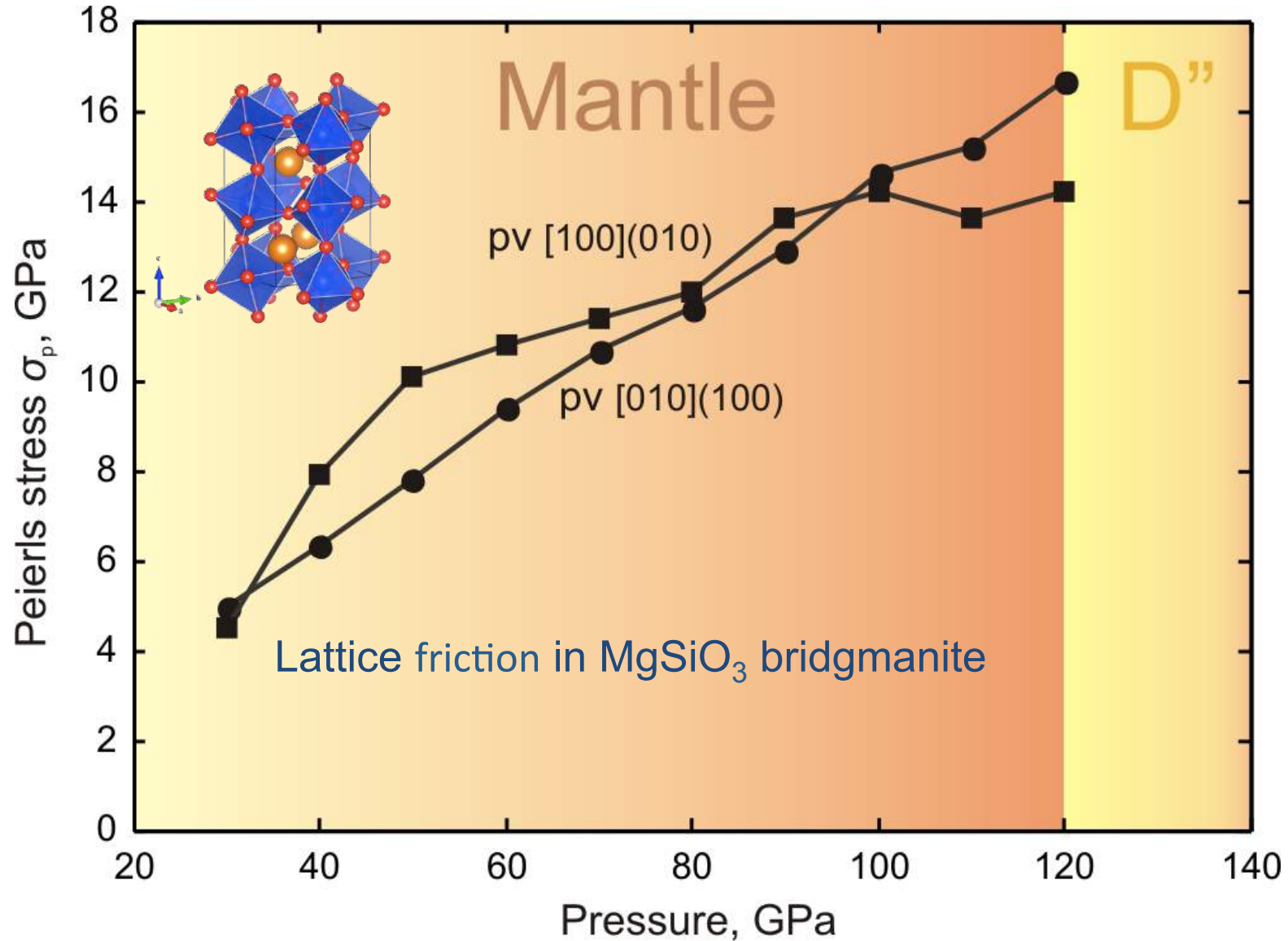
Received 20 January 2014; received in revised form 30 June 2014; accepted 2 July 2014

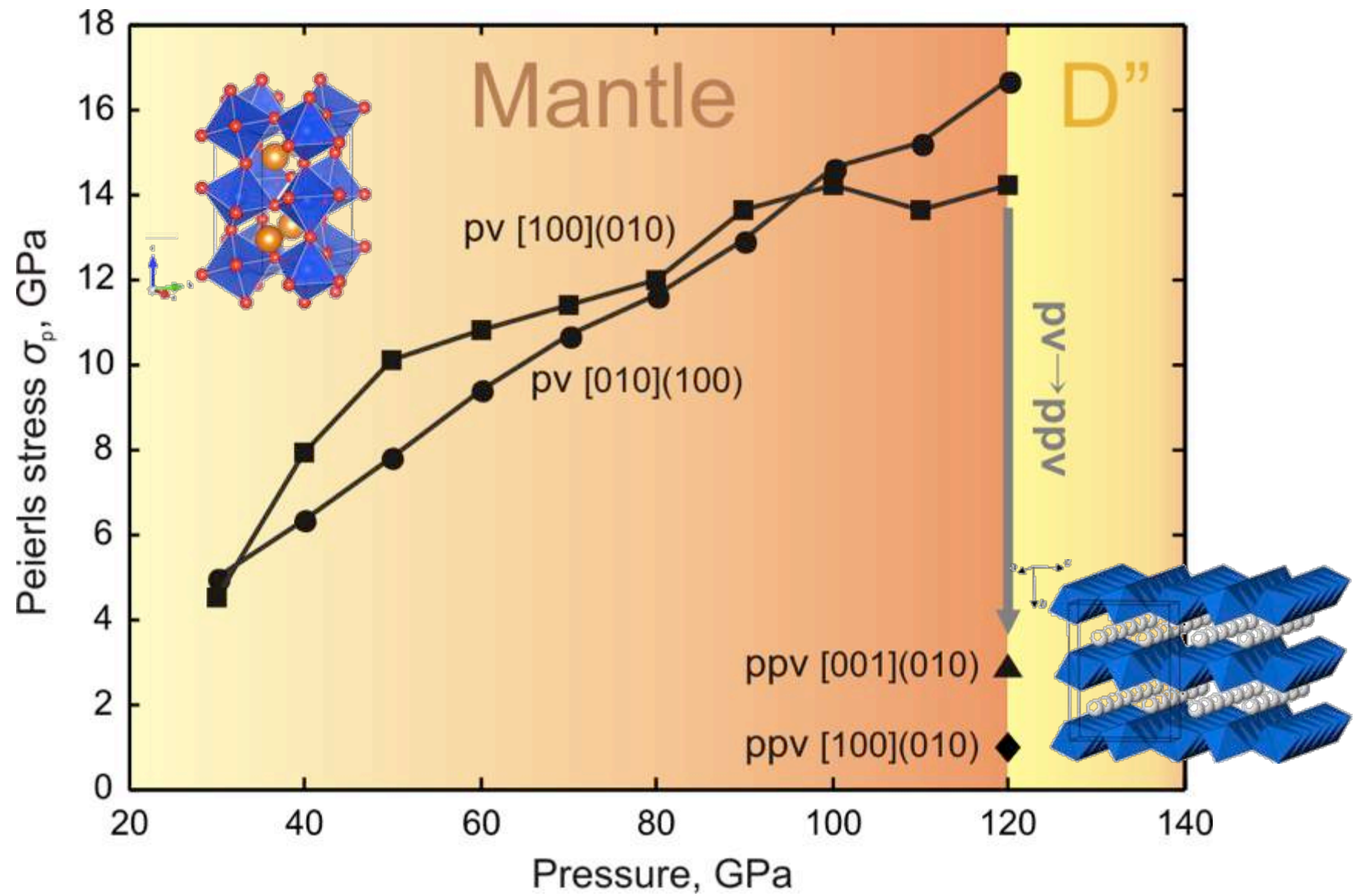


Peierls stress

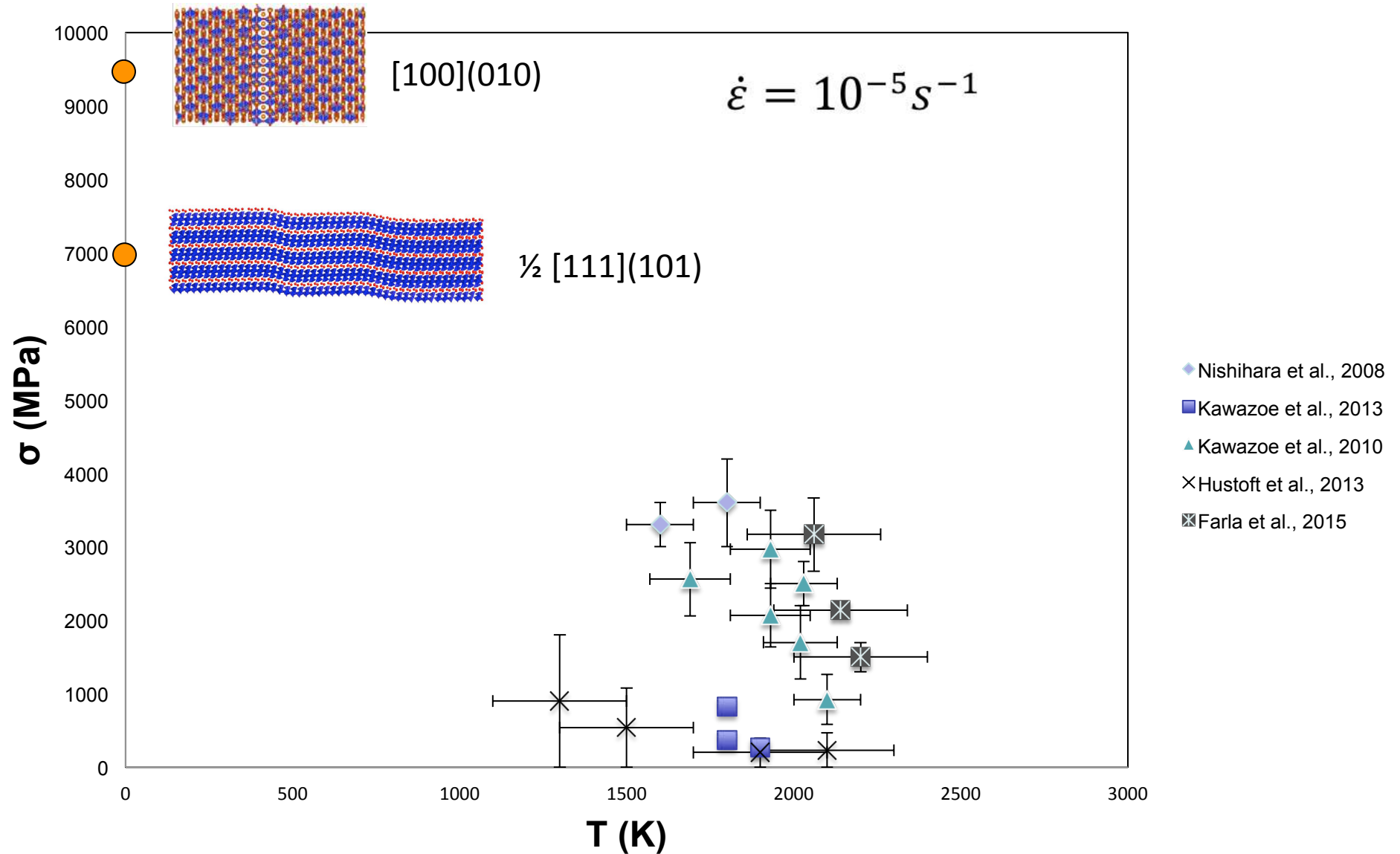


# Effect of Pressure on lattice friction: Peierls stress versus Pressure

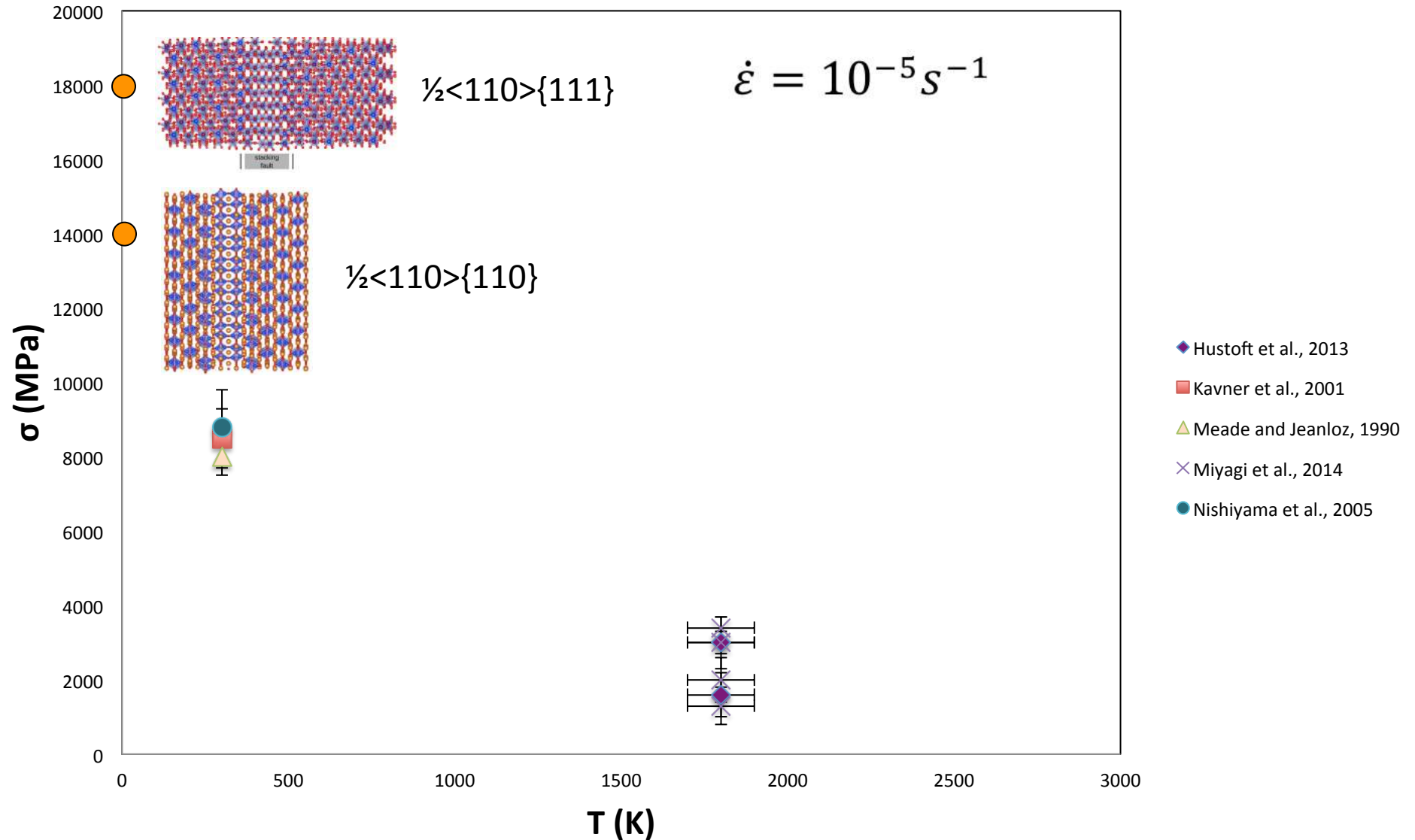




# Comparison with experimental data: Wadsleyite Peierls stresses



# Comparison with experimental data: Ringwoodite Peierls stresses

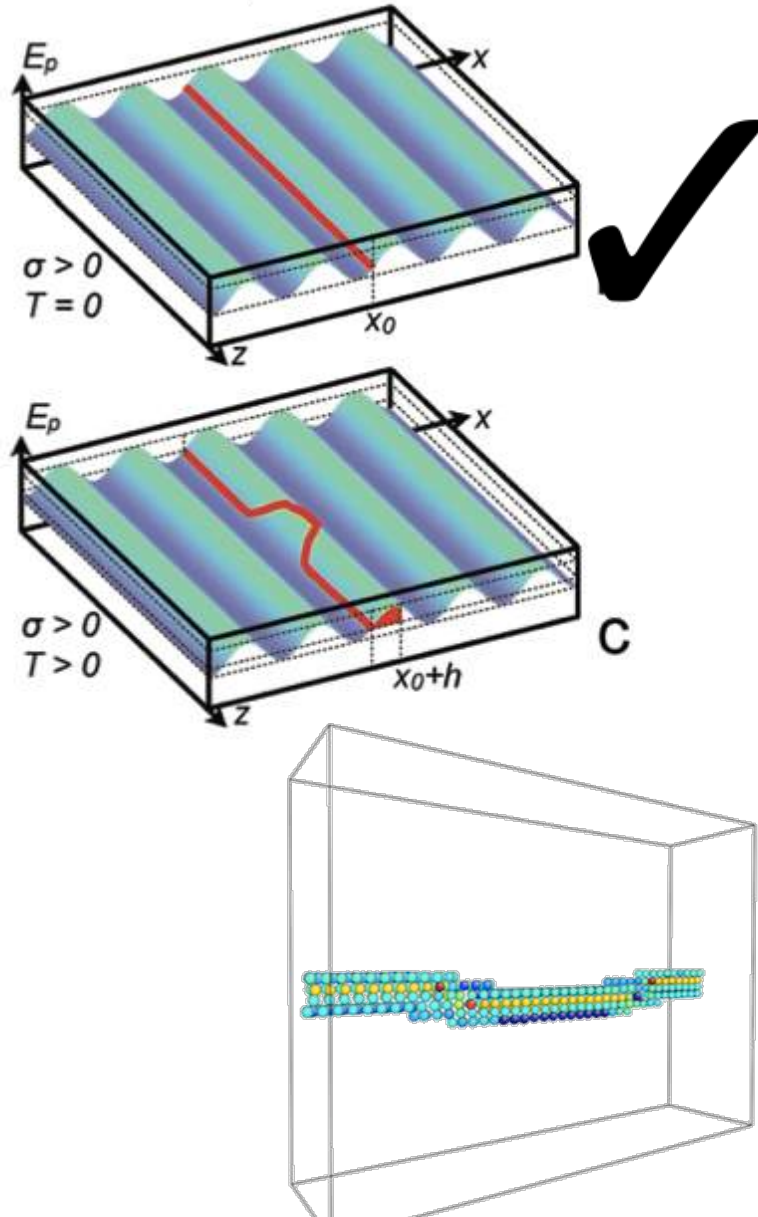




# Thermally activated glide velocity

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## KINK PAIR NUCLEATION AND CRITICAL SHEAR STRESS

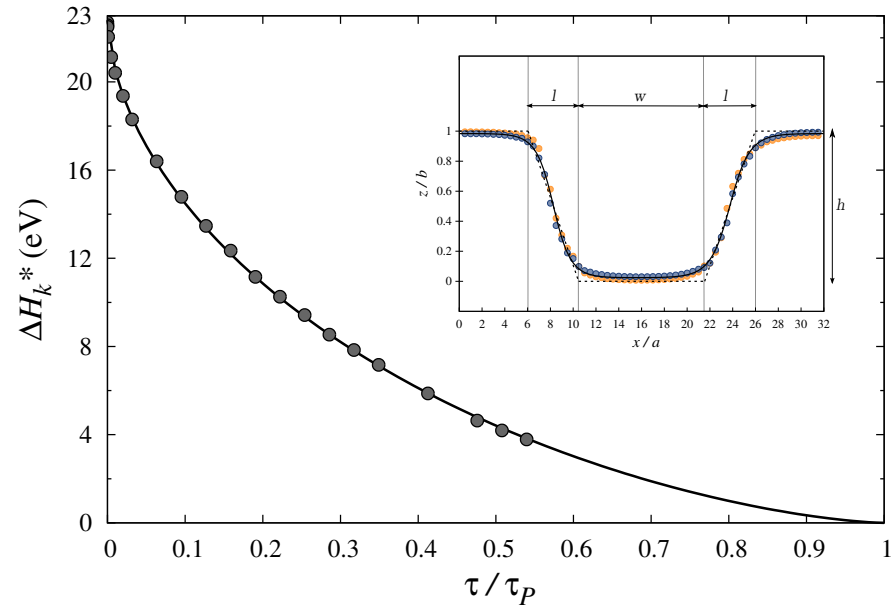
H. KOIZUMI,<sup>1</sup> H. O. K. KIRCHNER<sup>2</sup> and T. SUZUKI<sup>3</sup>

<sup>1</sup>Department of Physics, Meiji University, Higashimita, Tama-ku, Kawasaki 214, Japan,

<sup>2</sup>Institut de Sciences des Matériaux, Université Paris-Sud, F-91405 Orsay, France and

<sup>3</sup>Institute of Industrial Science, University of Tokyo, Roppongi, Minato-ku, Tokyo 106, Japan

$$\Delta H_k = \Delta E_{\text{elas}} + \Delta W_P - W_\tau$$



$$v = a' \cdot \frac{L}{w^*(\tau)} \cdot \frac{v_D b}{w^*(\tau)} \cdot \exp\left(-\frac{\Delta H^*(\tau)}{kT}\right)$$

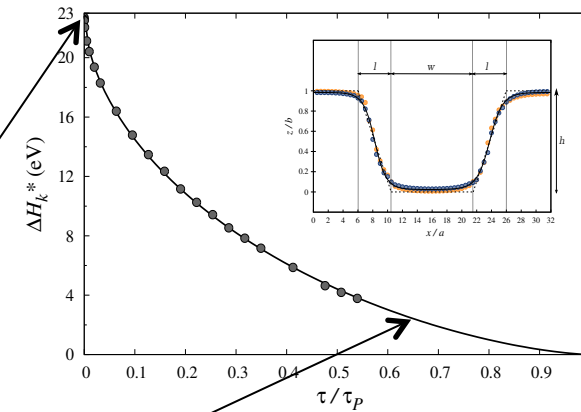
# From dislocations to plasticity: a practical example

Orowan equation  $\dot{\epsilon} = \rho b v(\tau, T)$

Kraych et al. EPSL 2016

$$v(\tau, T) = v_0 \exp\left(\frac{-\Delta H^*(\tau)}{kT}\right)$$

$$\Delta H^*(\tau) = 2H_k \left(1 - \left(\frac{\tau}{\tau_p}\right)^p\right)^q$$



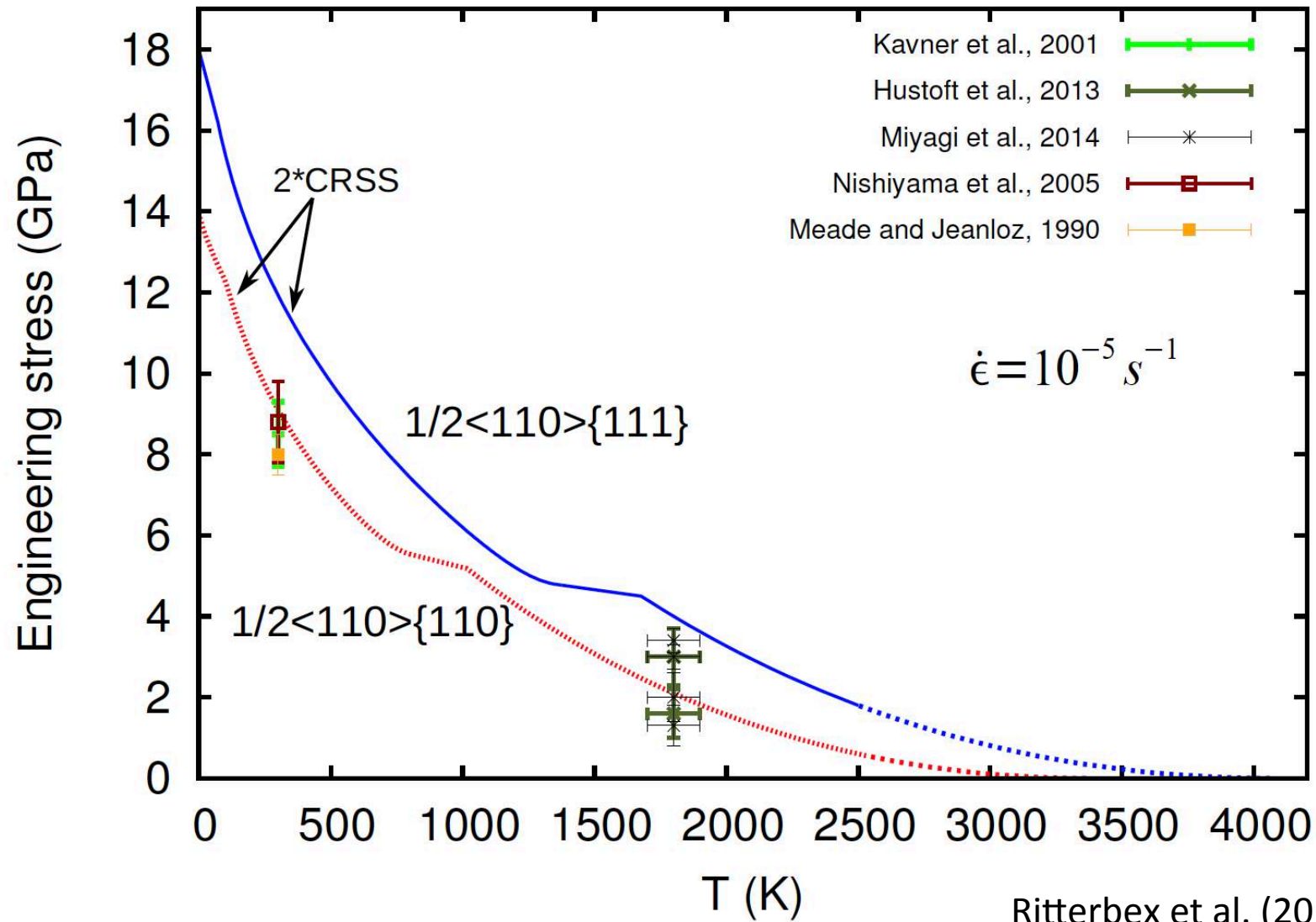
$$\tau = \tau_p \left(1 - \left(C \frac{T}{H_k}\right)^{1/q}\right)^{1/p}$$

$$C = \frac{k}{2} \ln \left( \frac{v_D a' b^2 \sqrt{\rho}}{2w^* \dot{\epsilon}} \right)$$

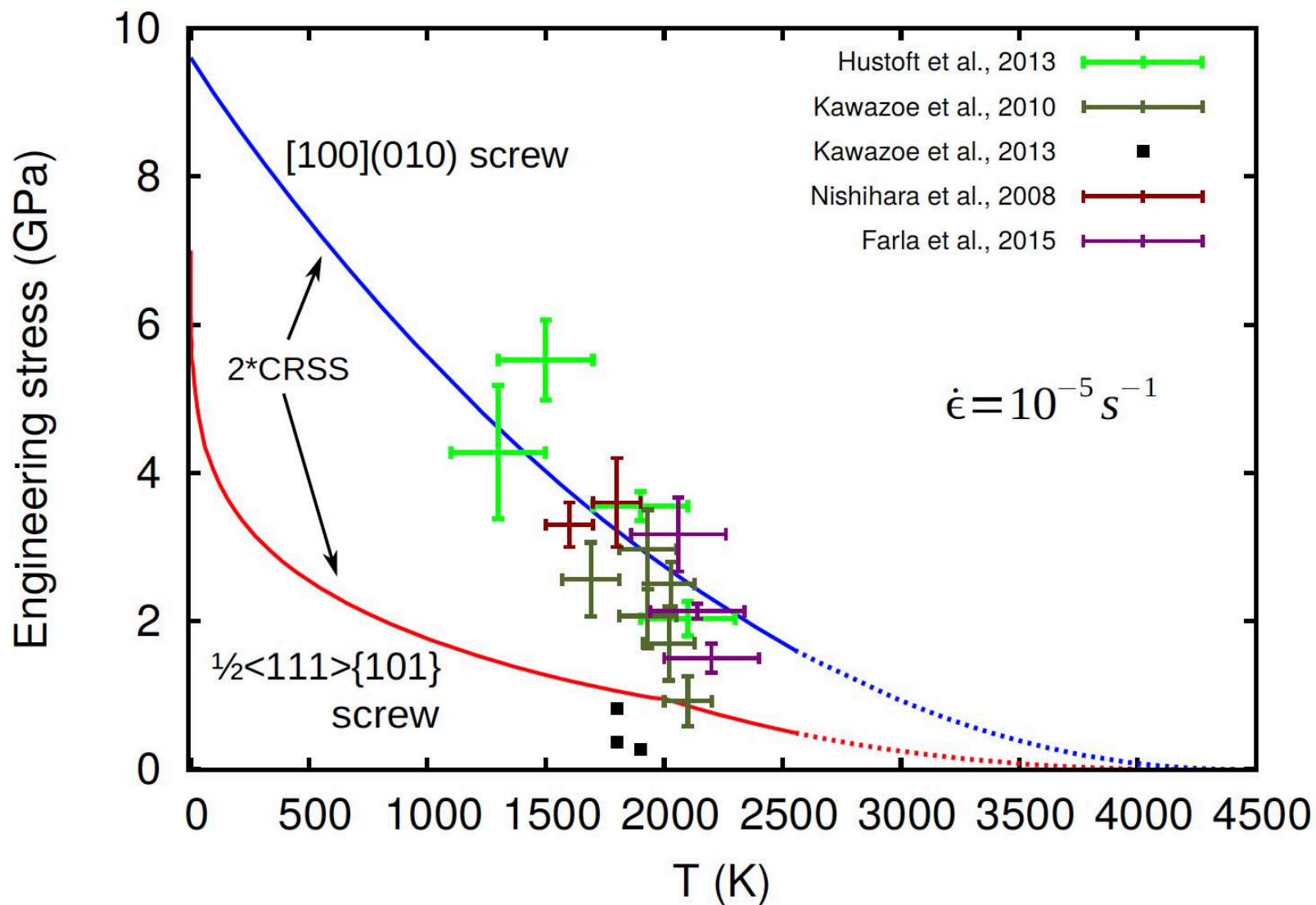


The strain rate is here

# Comparison with experimental data: Ringwoodite (20 GPa)

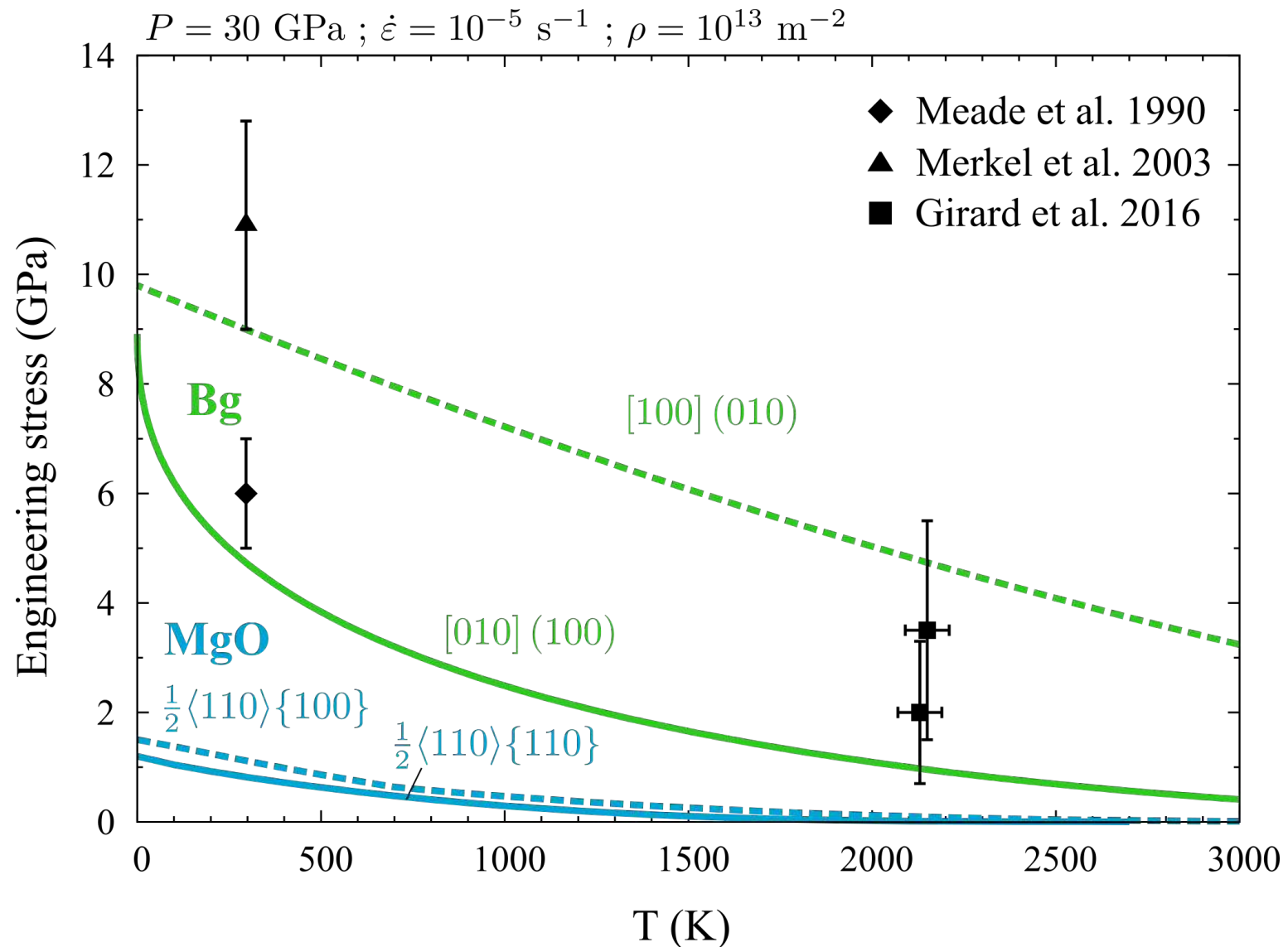


# Dislocation glide in wadsleyite (15 GPa)



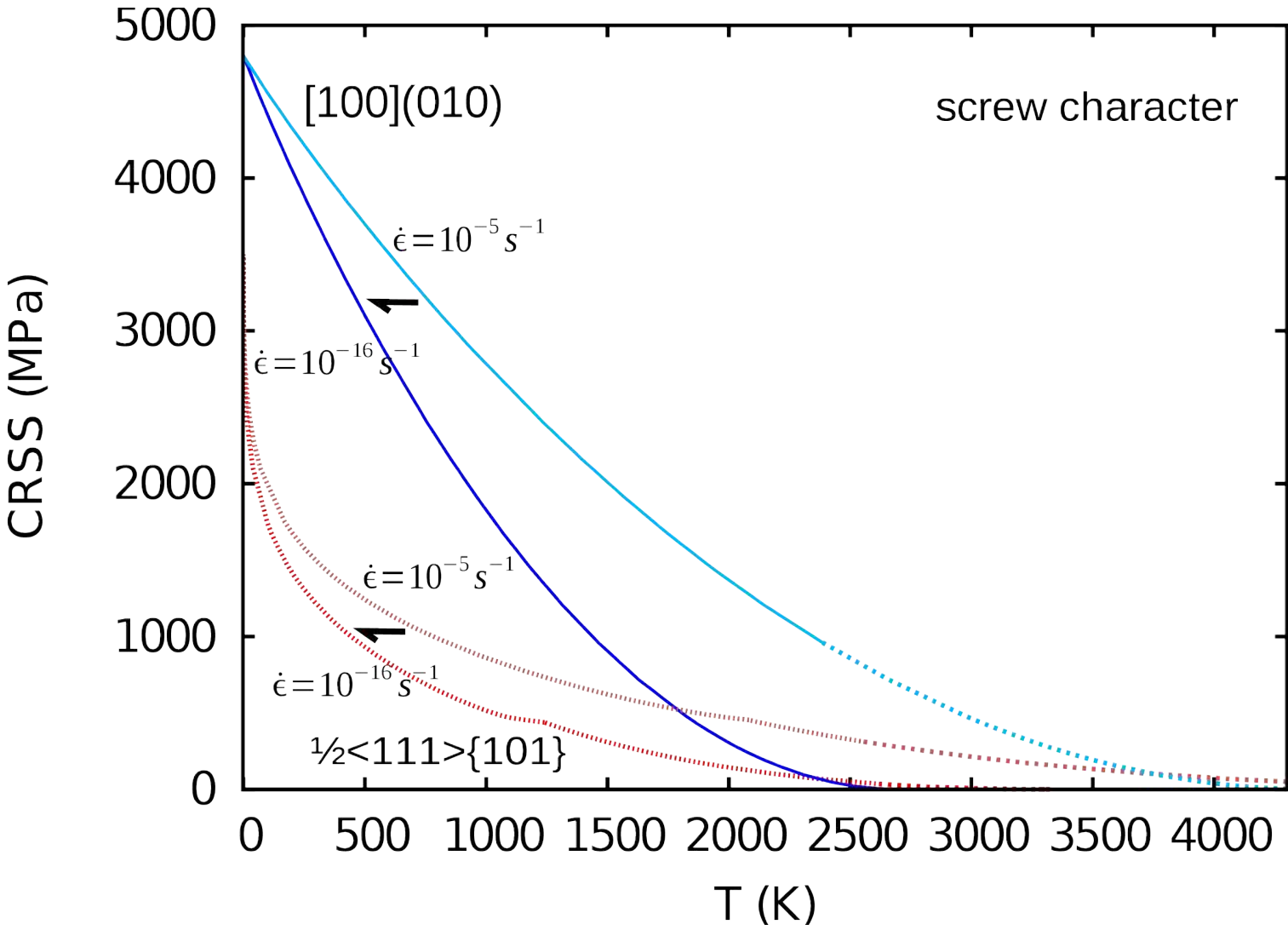


# Dislocation glide in bridgmanite (30 GPa)



# Dislocation glide in wadsleyite: from the lab to the mantle

Decreasing strain rate shifts CRSS to lower stress values



# Dislocation glide in wadsleyite: from the lab to the mantle

Decreasing strain rate shifts CRSS to lower stress values

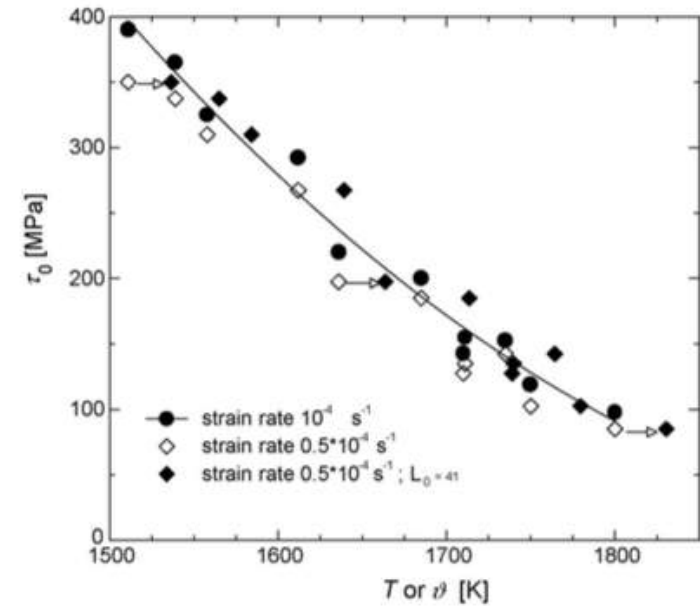
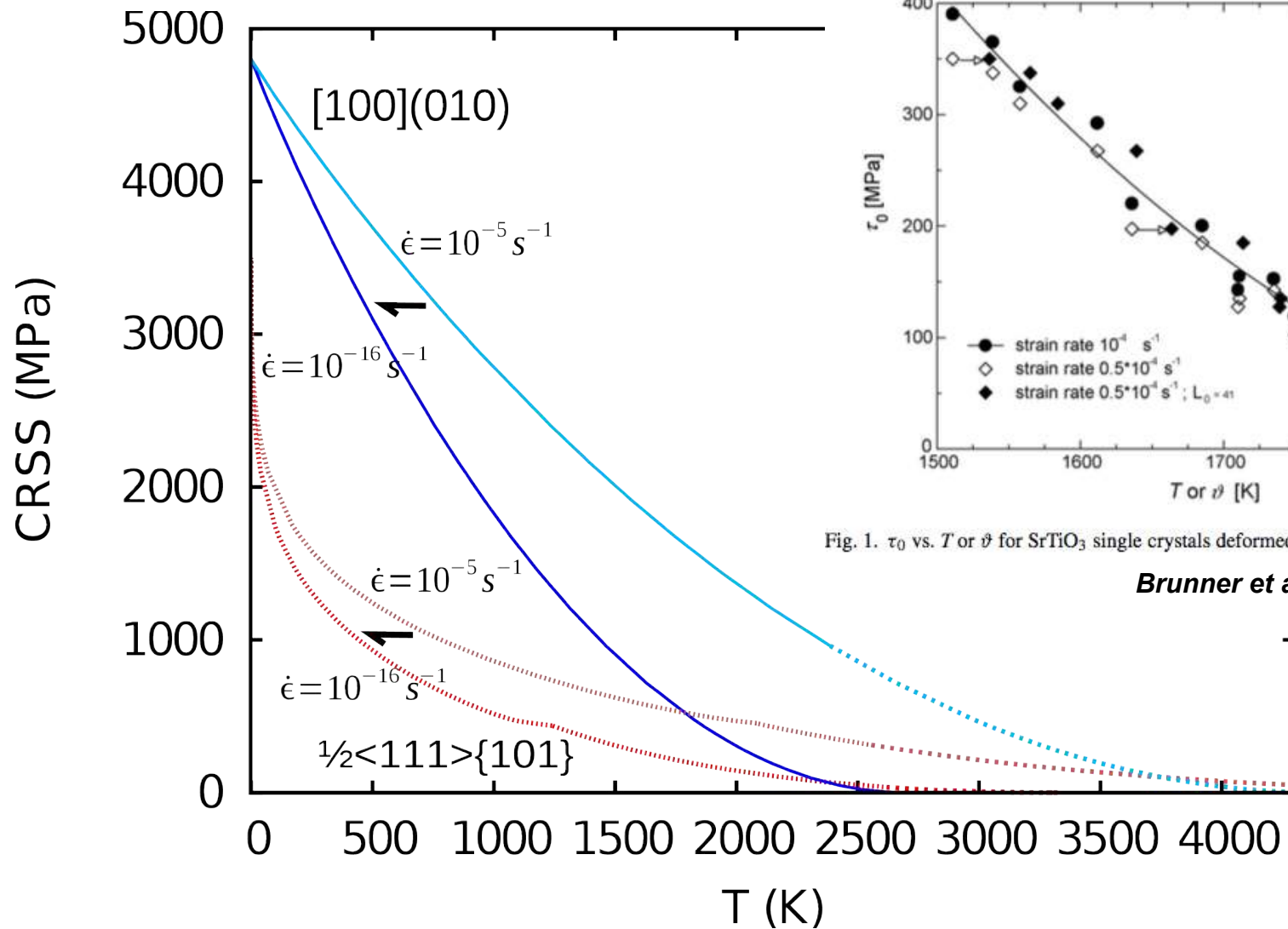
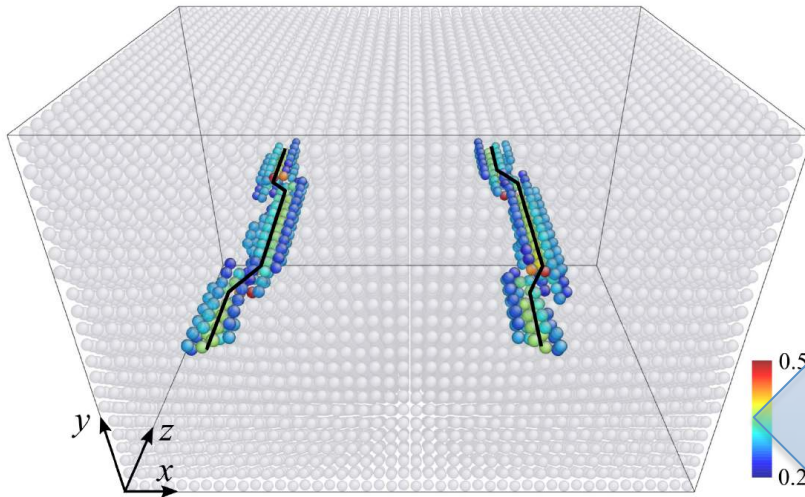


Fig. 1.  $\tau_0$  vs.  $T$  or  $\vartheta$  for  $\text{SrTiO}_3$  single crystals deformed at high temperatures.

*Brunner et al. (2008)*

# Multi-Scale model of dislocation creep

Atomistic calculations of dislocation core properties

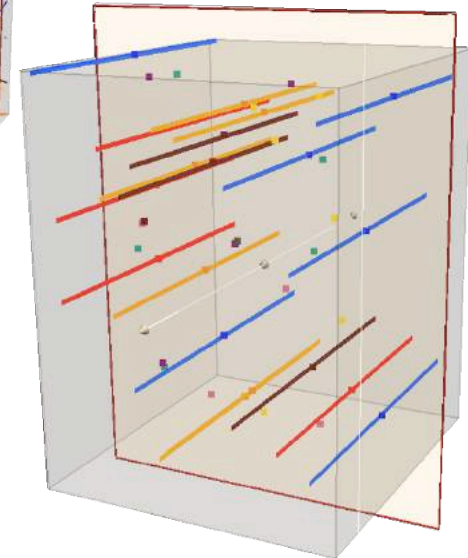


Thermally activated glide velocity

Dislocation Dynamic (DD) simulation



Simple framework to introduce climb  
2.5-D DD LEM/CNRS-ONERA in Paris



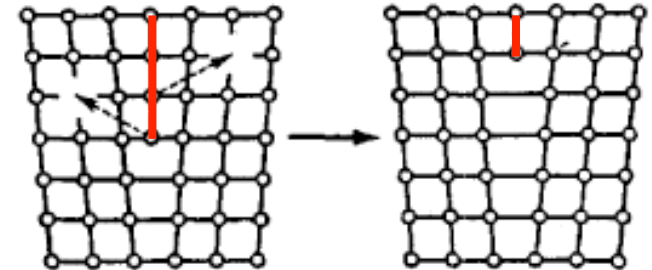
Climb velocity





# Climb velocity in Dislocation Dynamics simulations

- Point defects (vacancies) diffuse toward the dislocations
- They are absorbed (or emitted)

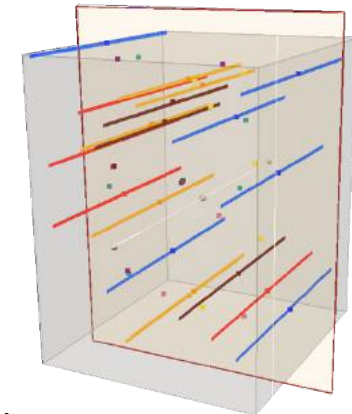


The dislocation moves away from its glide plane

$$v_{climb} = \frac{2\pi}{\ln R/r_c} \frac{D_{Si}^{sd}}{b} \left( \exp\left(\frac{\tau\Omega}{kT}\right) - 1 \right)$$

Steady state conditions

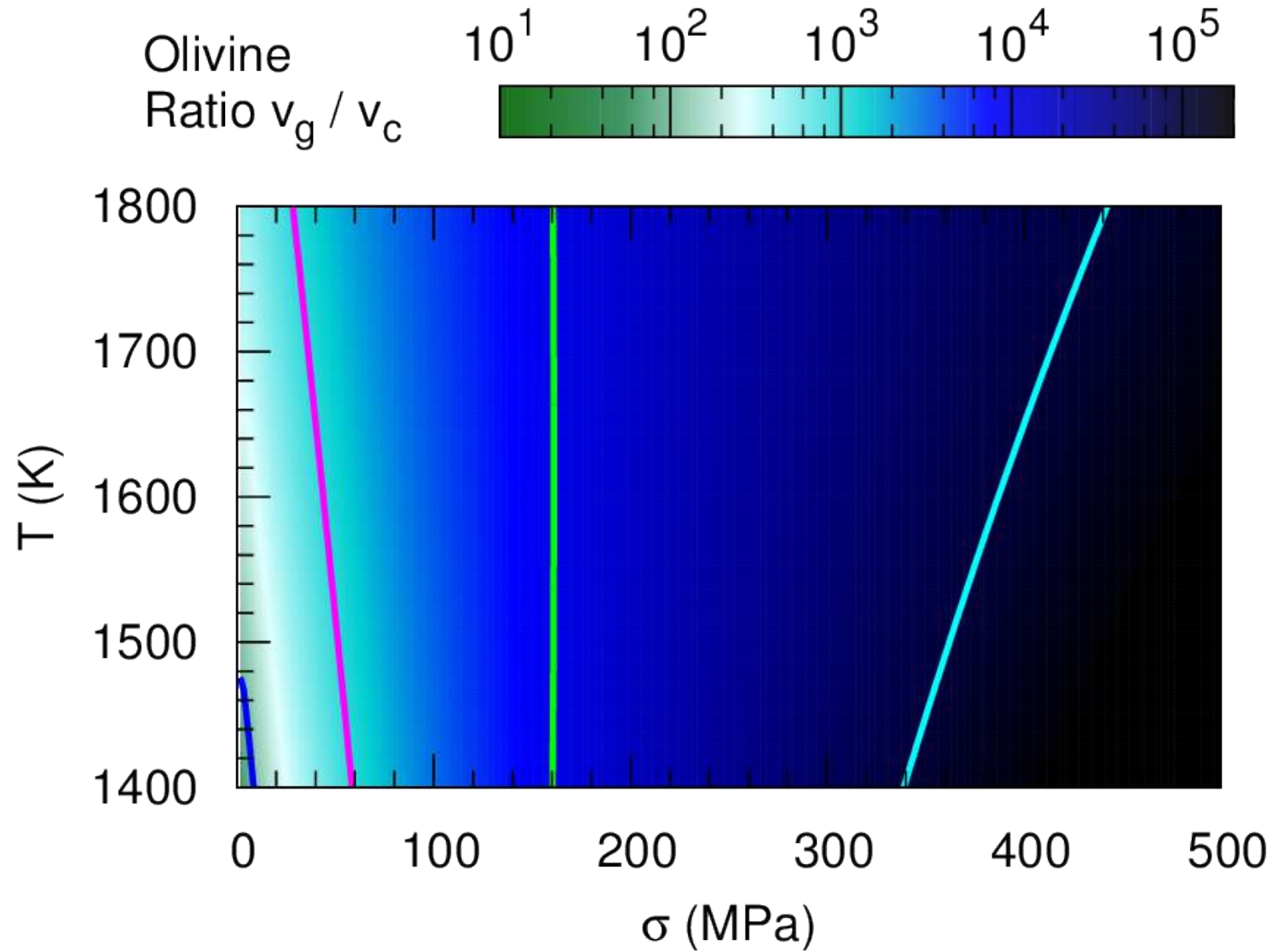
Constant vacancy concentration



(see Mordehai et al. Phil. Mag. 2008 or Keralavarma et al. 2012)

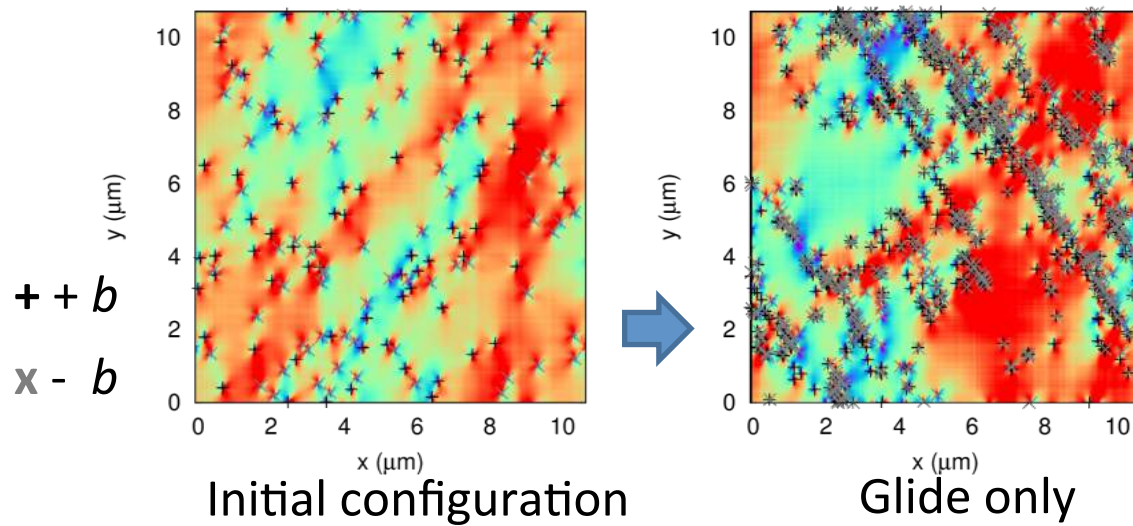
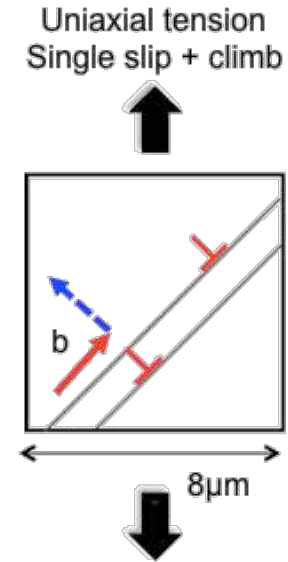
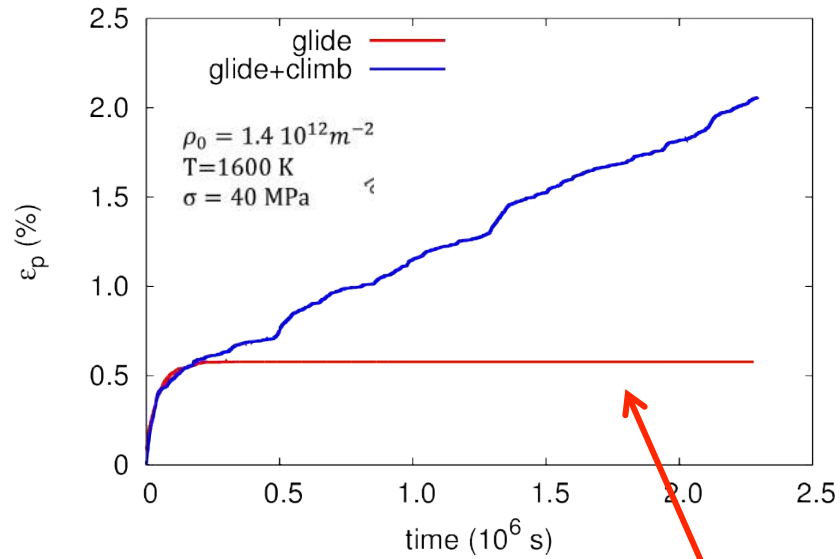
# Interplay between glide and climb

[100] dislocations



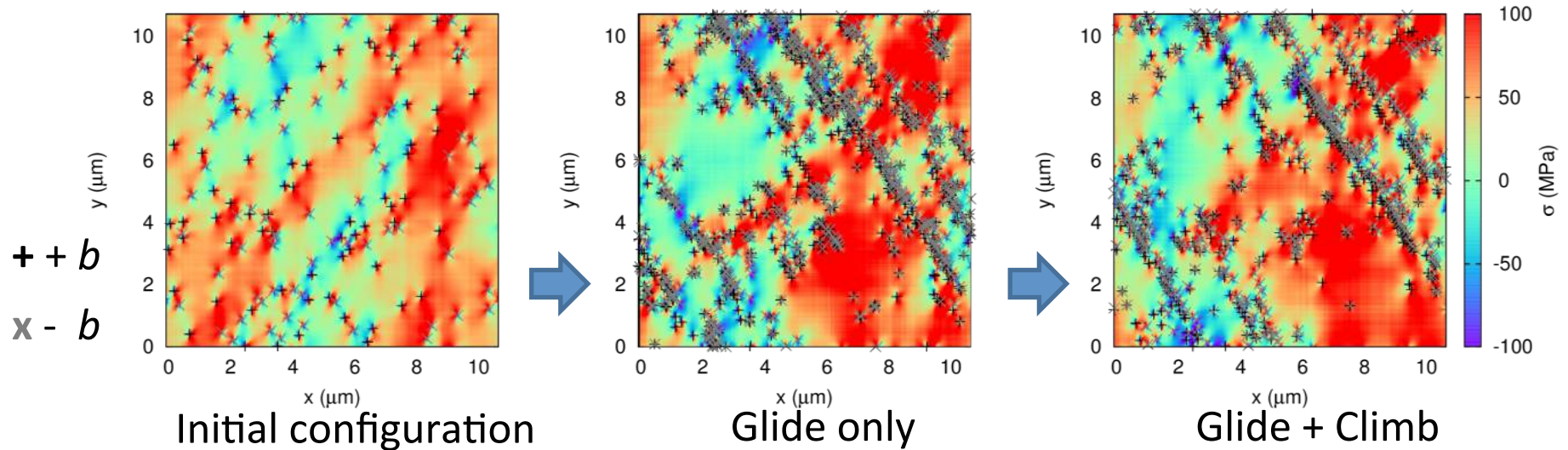
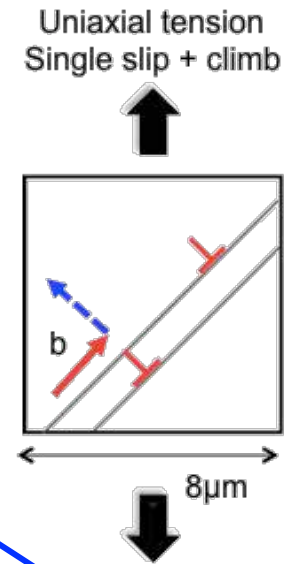
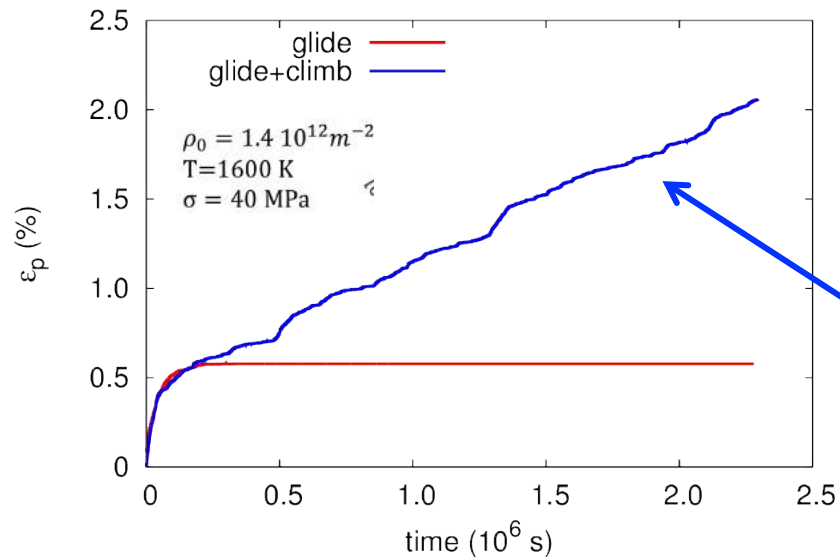
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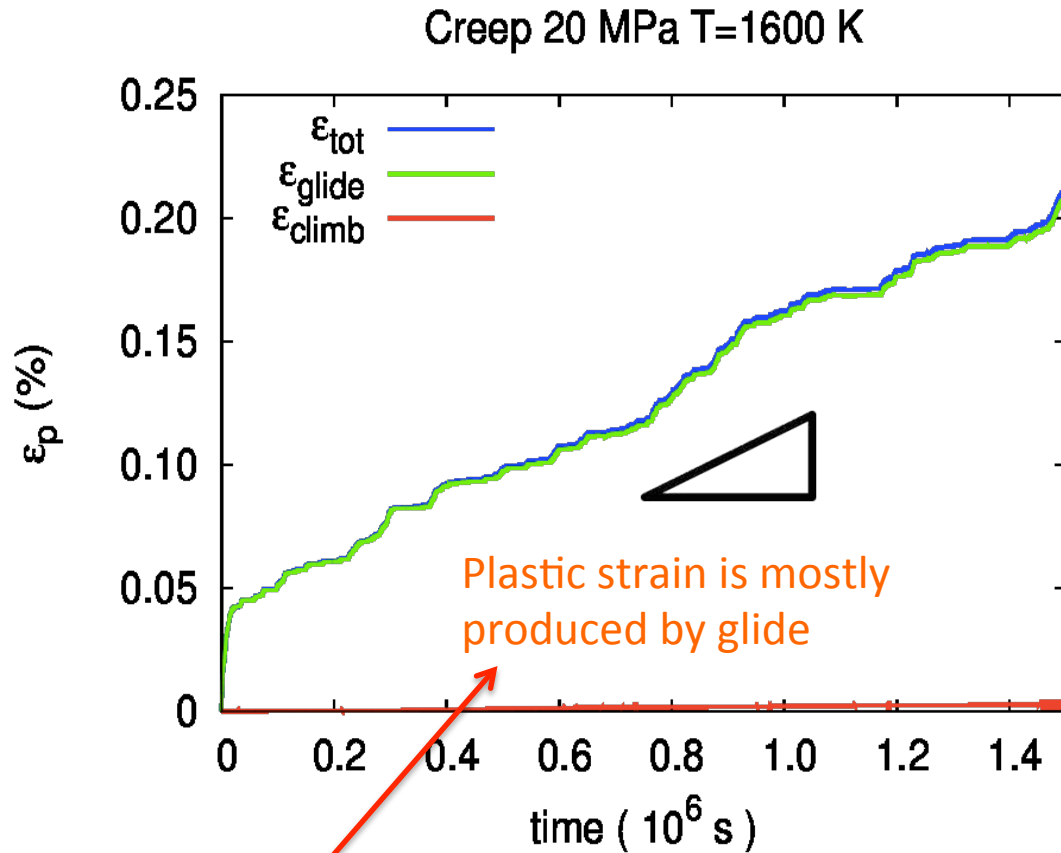
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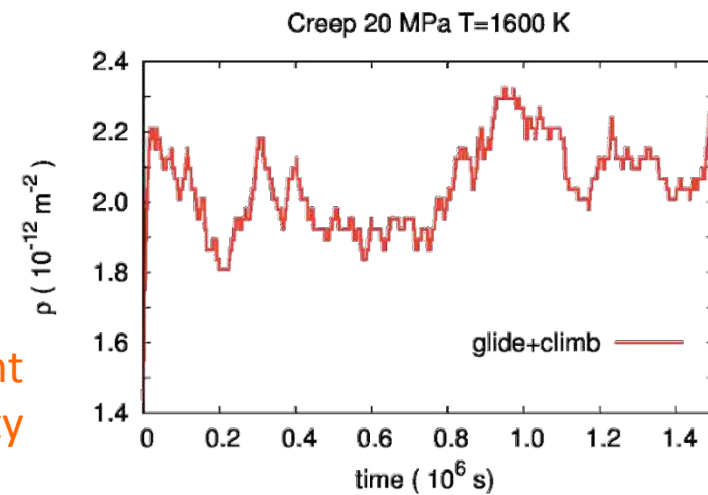
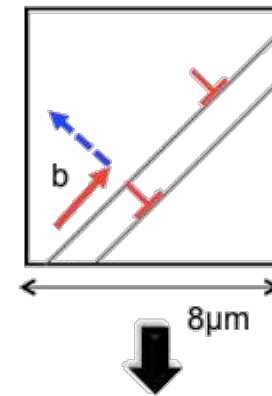
[100] dislocations



Weertman  
creep

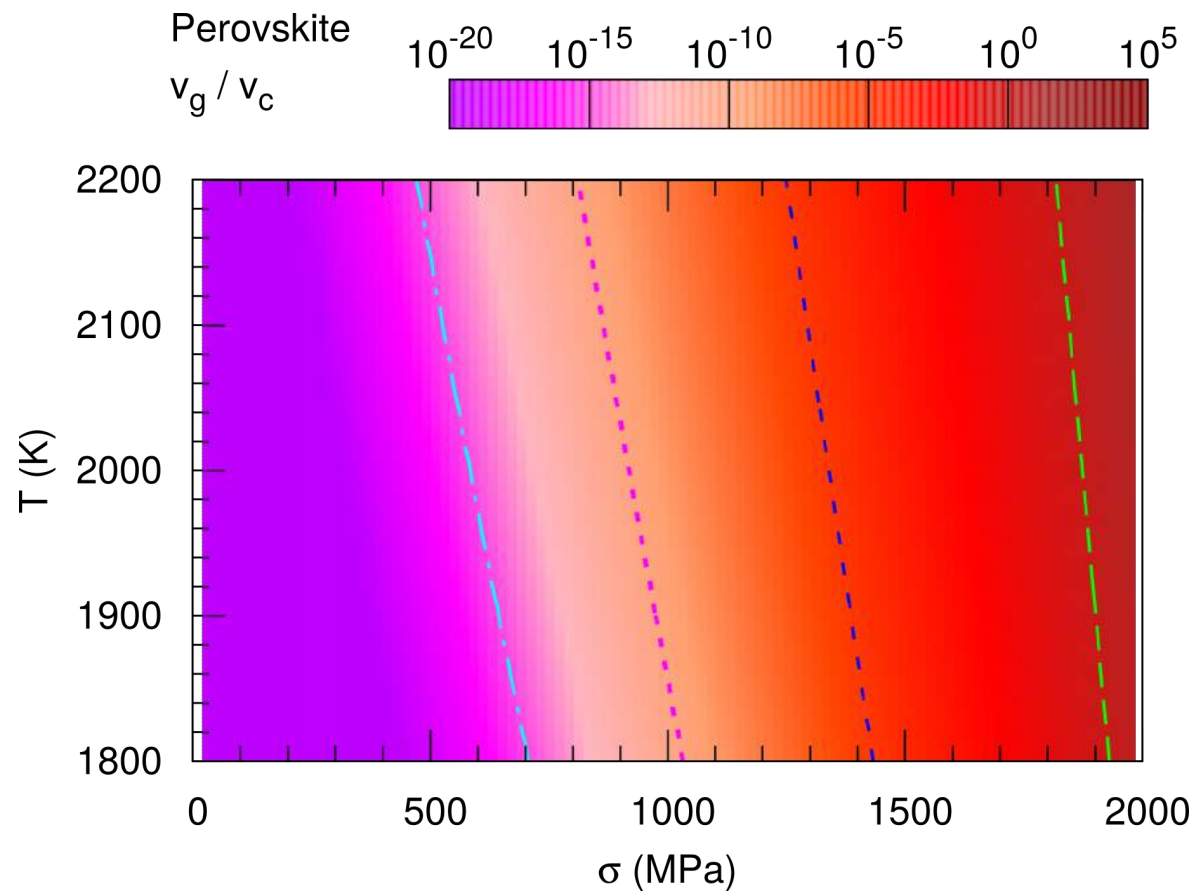
At steady state: nearly constant  
dislocation density

Uniaxial tension  
Single slip + climb

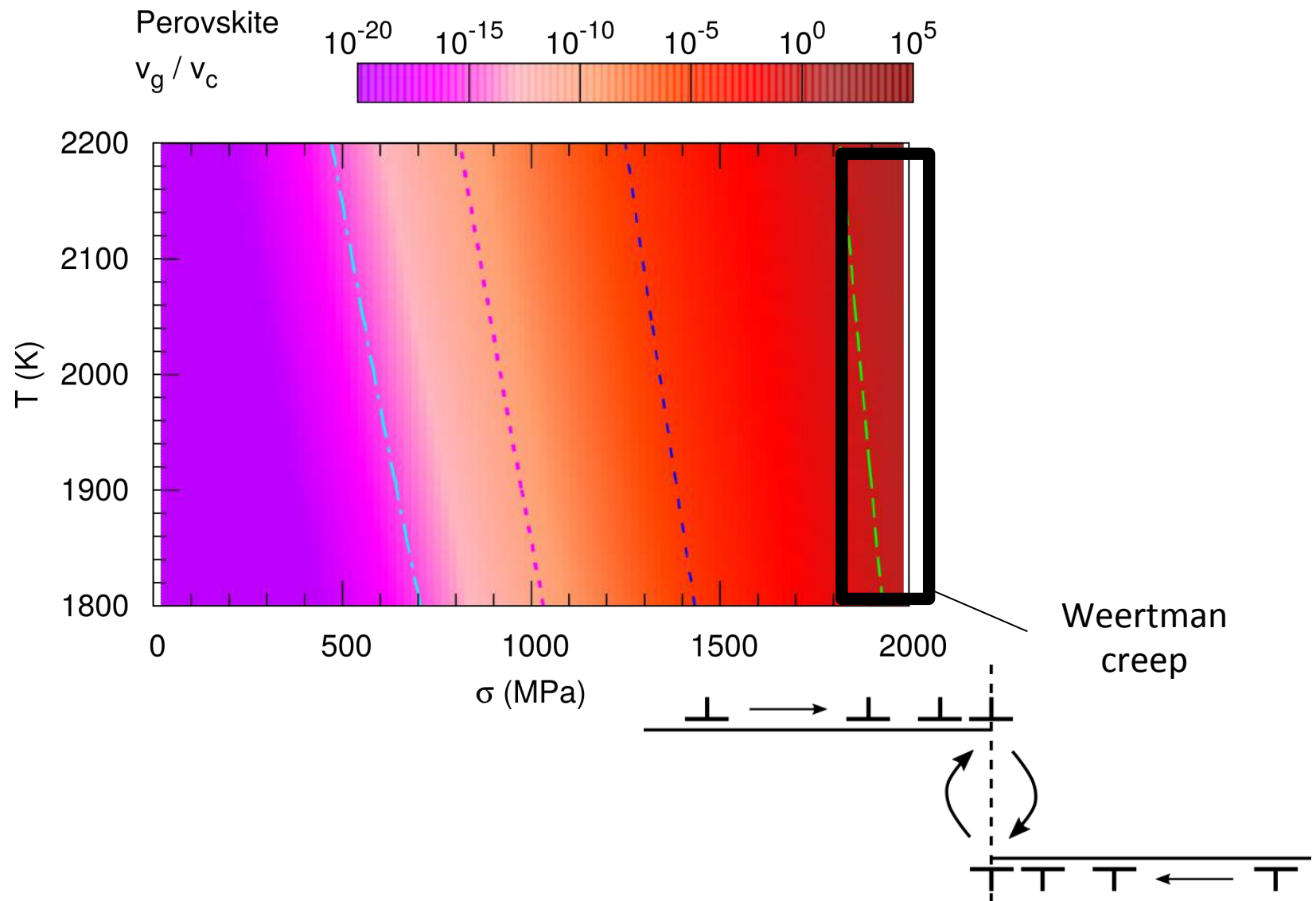




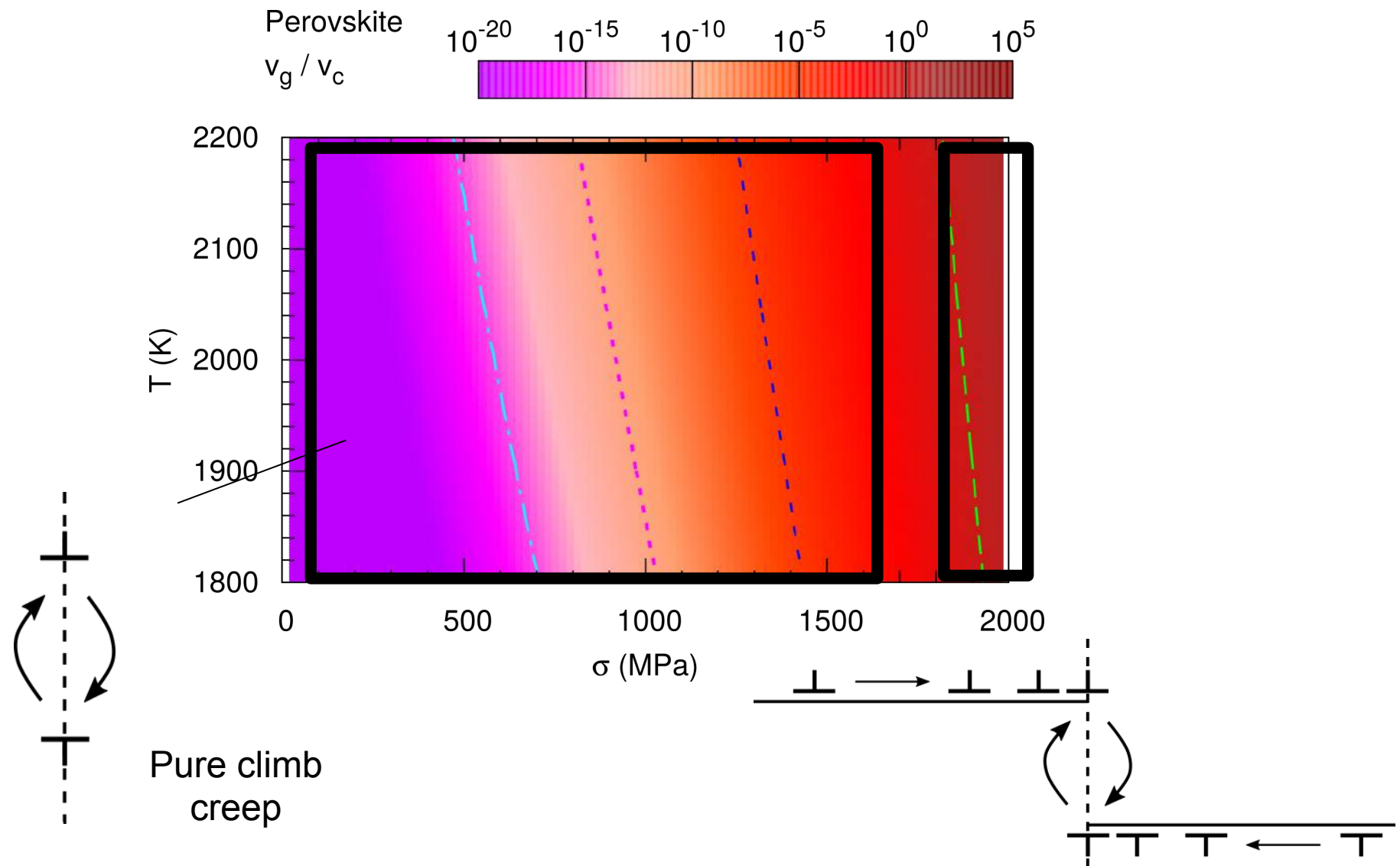
# Glide versus Climb in MgSiO<sub>3</sub> perovskite as a function of Stress and Temperature



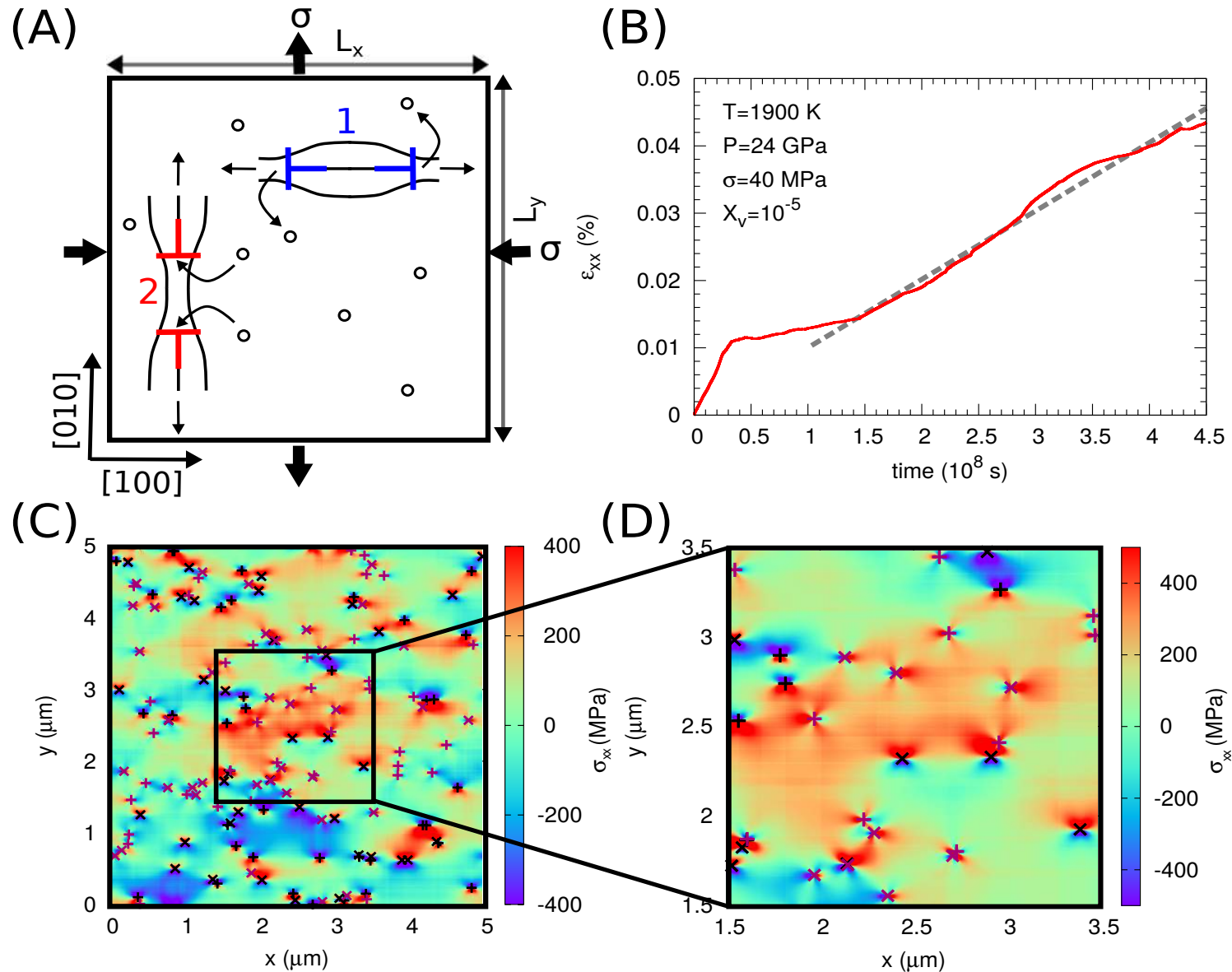
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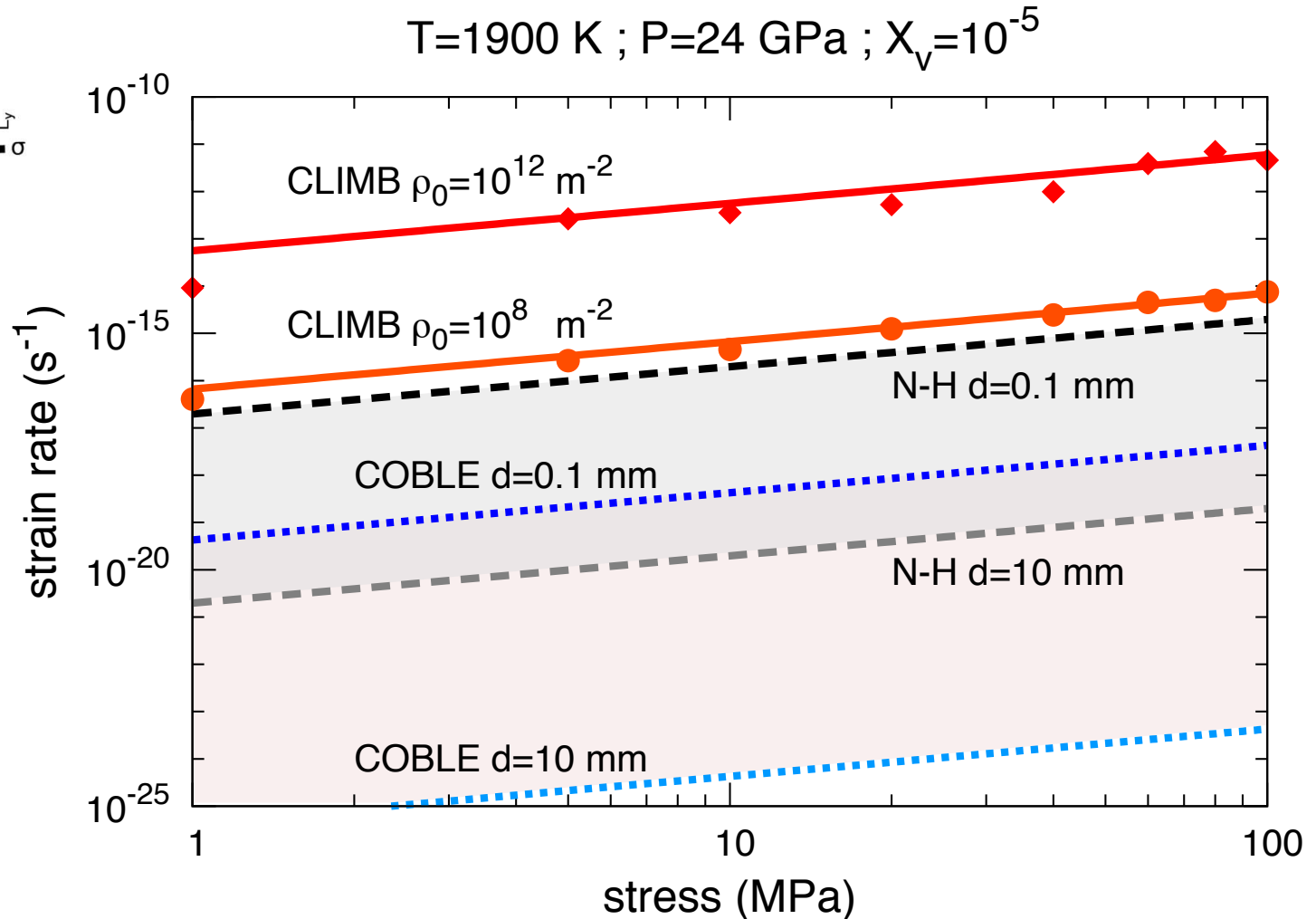
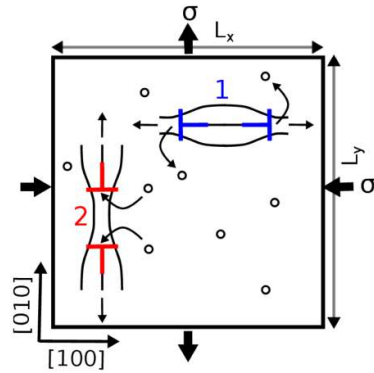
# Glide versus Climb in MgSiO<sub>3</sub> perovskite as a function of Stress and Temperature



# 2.5 DD simulations of pure climb creep



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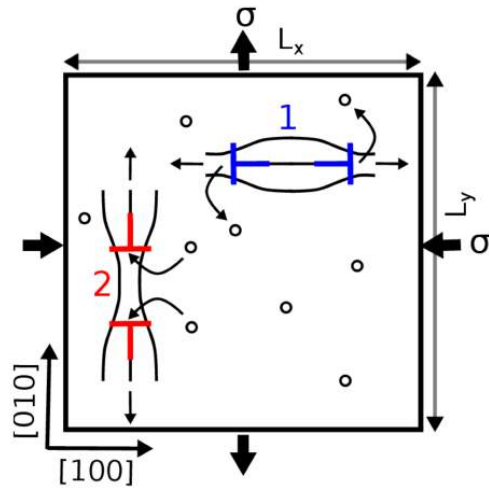


$d > 0,1 \text{ mm} \rightarrow$  Pure climb creep  $>$  NH creep



# Pure climb creep:

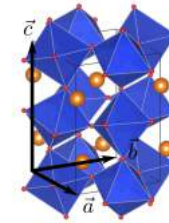
A very important mechanism for planetary interiors rheology



A few facts:

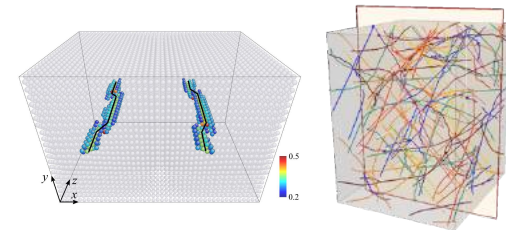
- Strain is produced by dislocation **climb**
- Strain is not produced by shear: no crystal preferred orientations
- No grain size dependence
- Controlled by diffusion, but rheology *may not* be linear

# Conclusions



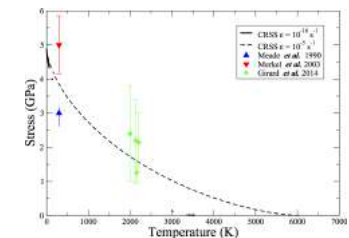
We develop a multi-scale model of dislocation creep in high pressure mineral

- Combining atomic scale and meso-scale DD simulations

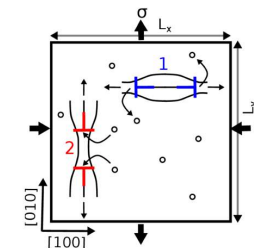


- Calculated glide properties are found in agreement with experiments

- Confirm that at relatively low stress glide is highly prohibited



- Creep may involves pure climb ( grain size > a few mm)



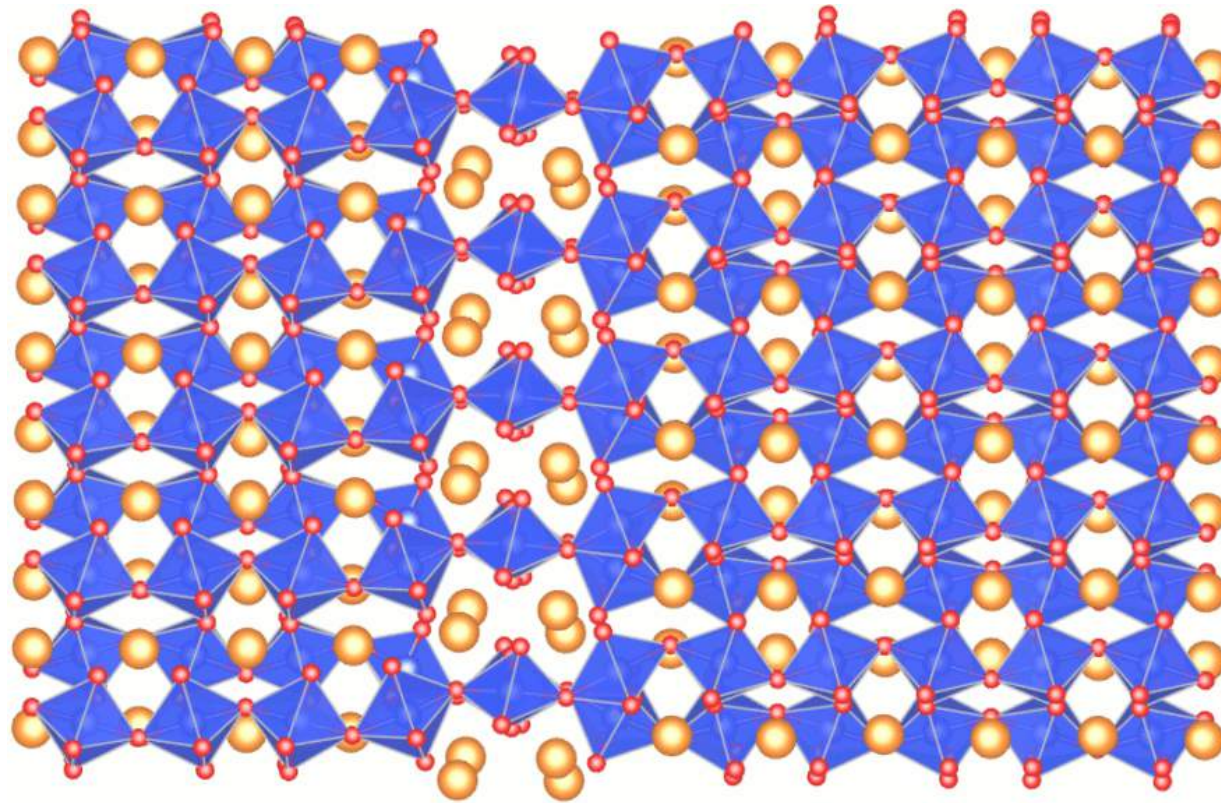
Thanks for your attention

This work was supported by funding from the European Research Council under the Seventh Framework Programme (FP7), ERC grant N°290424 – RheoMan.

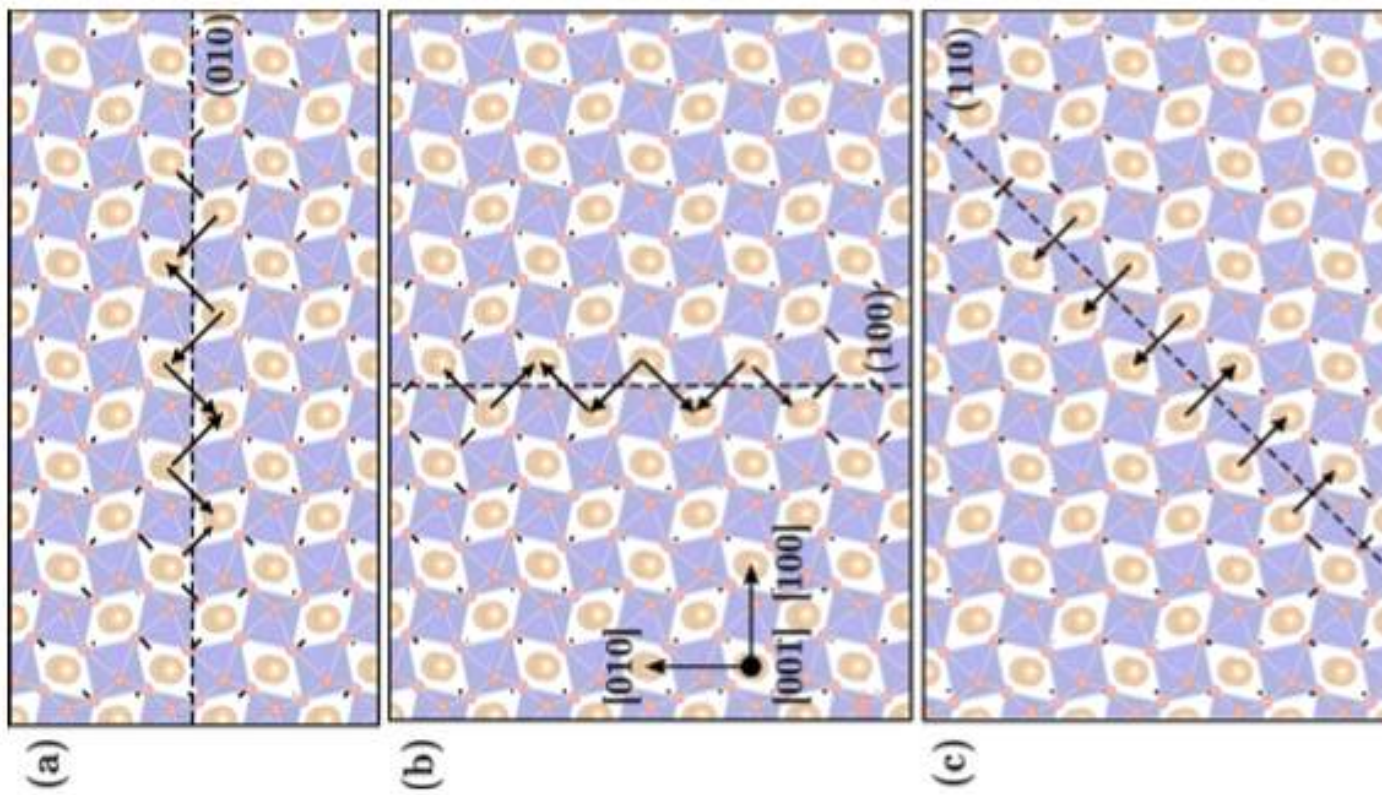
[www.rheoman.eu](http://www.rheoman.eu)



atomistic calculation of thermally activated glide velocity



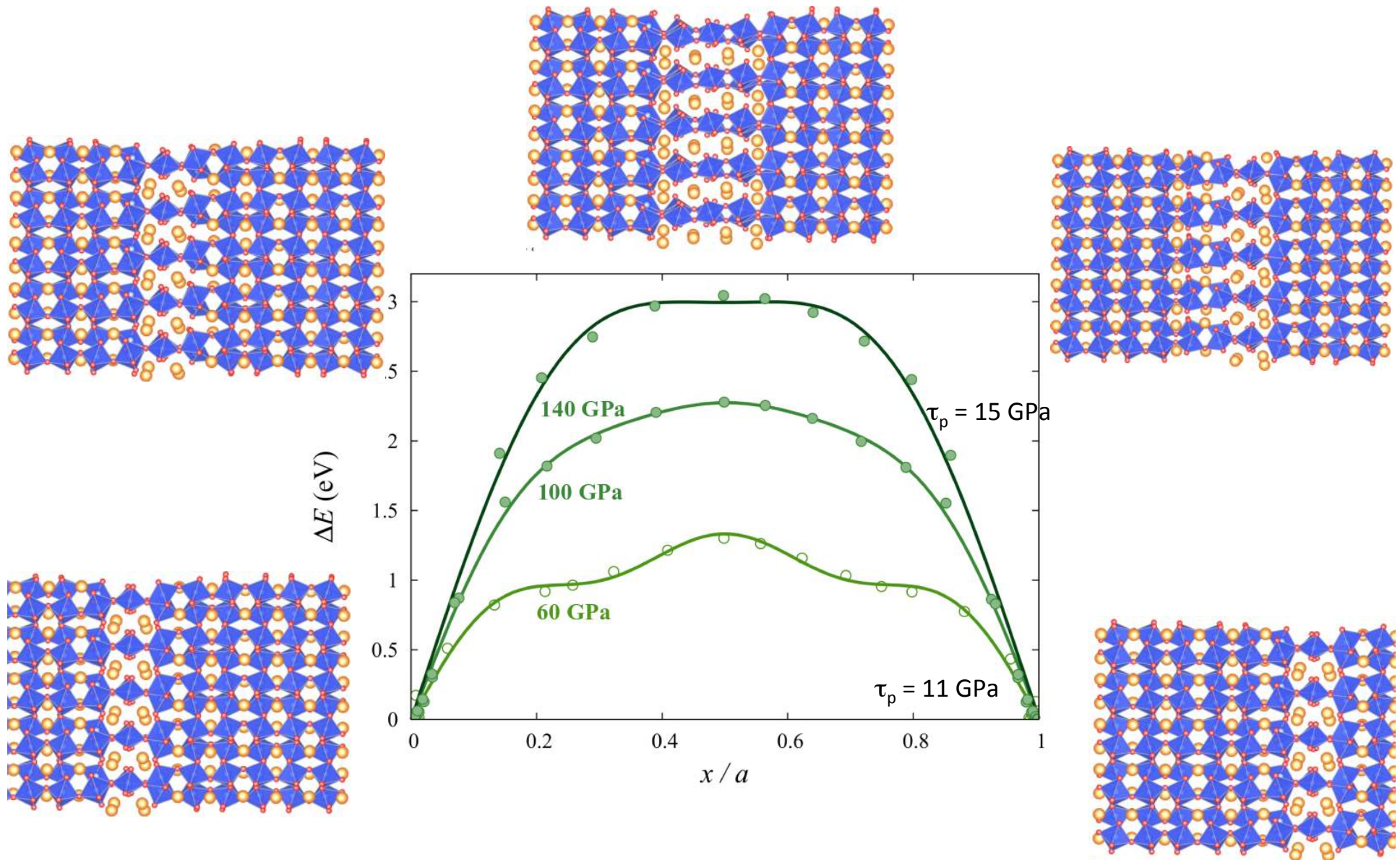




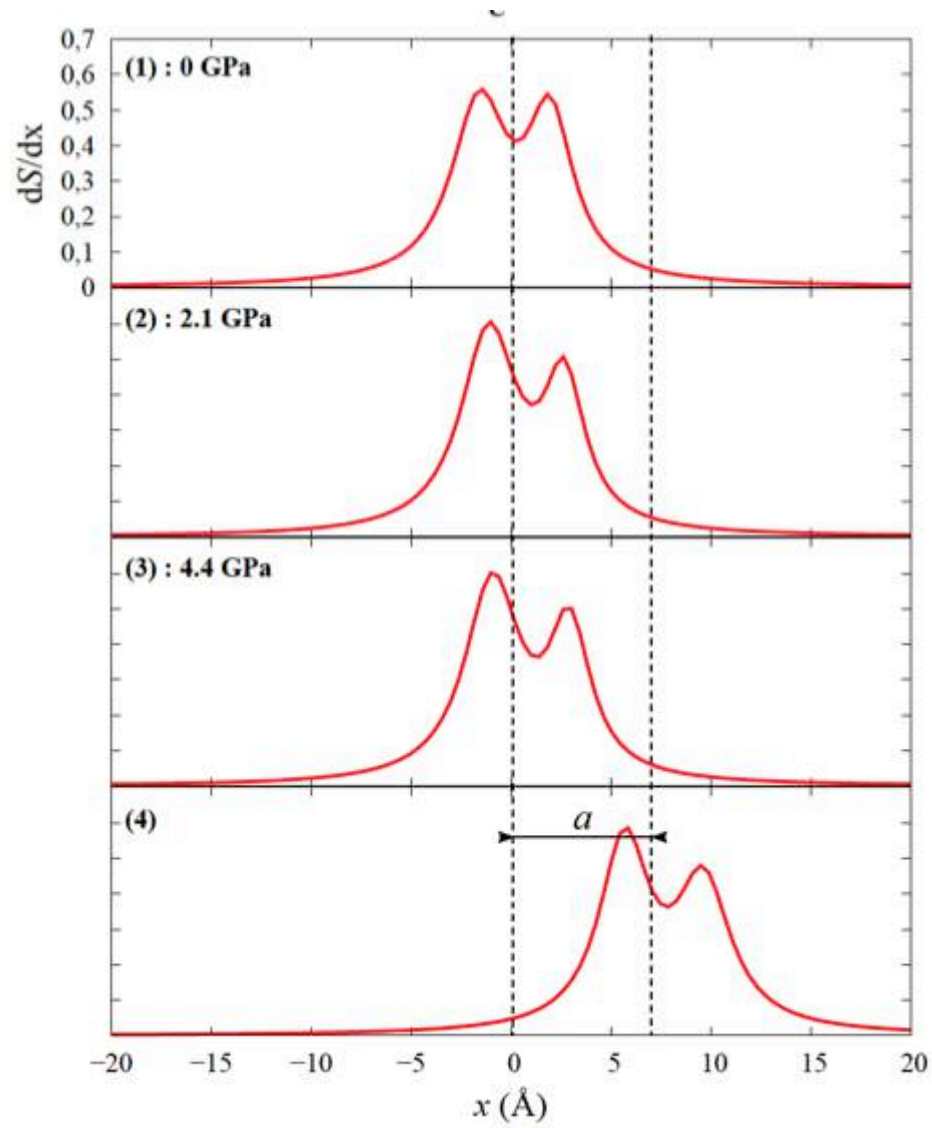
[001] dislocation in MgSiO<sub>3</sub> Pv

# Peierls potential computed using NEB

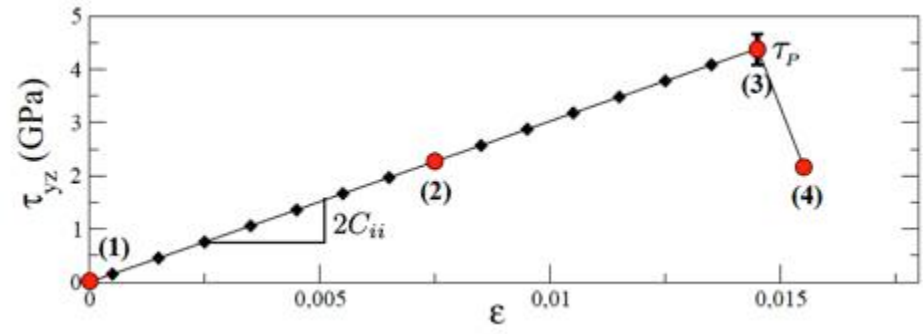
## Example of screw [010](100) dislocation



# Dislocation core structure and Peierls stress

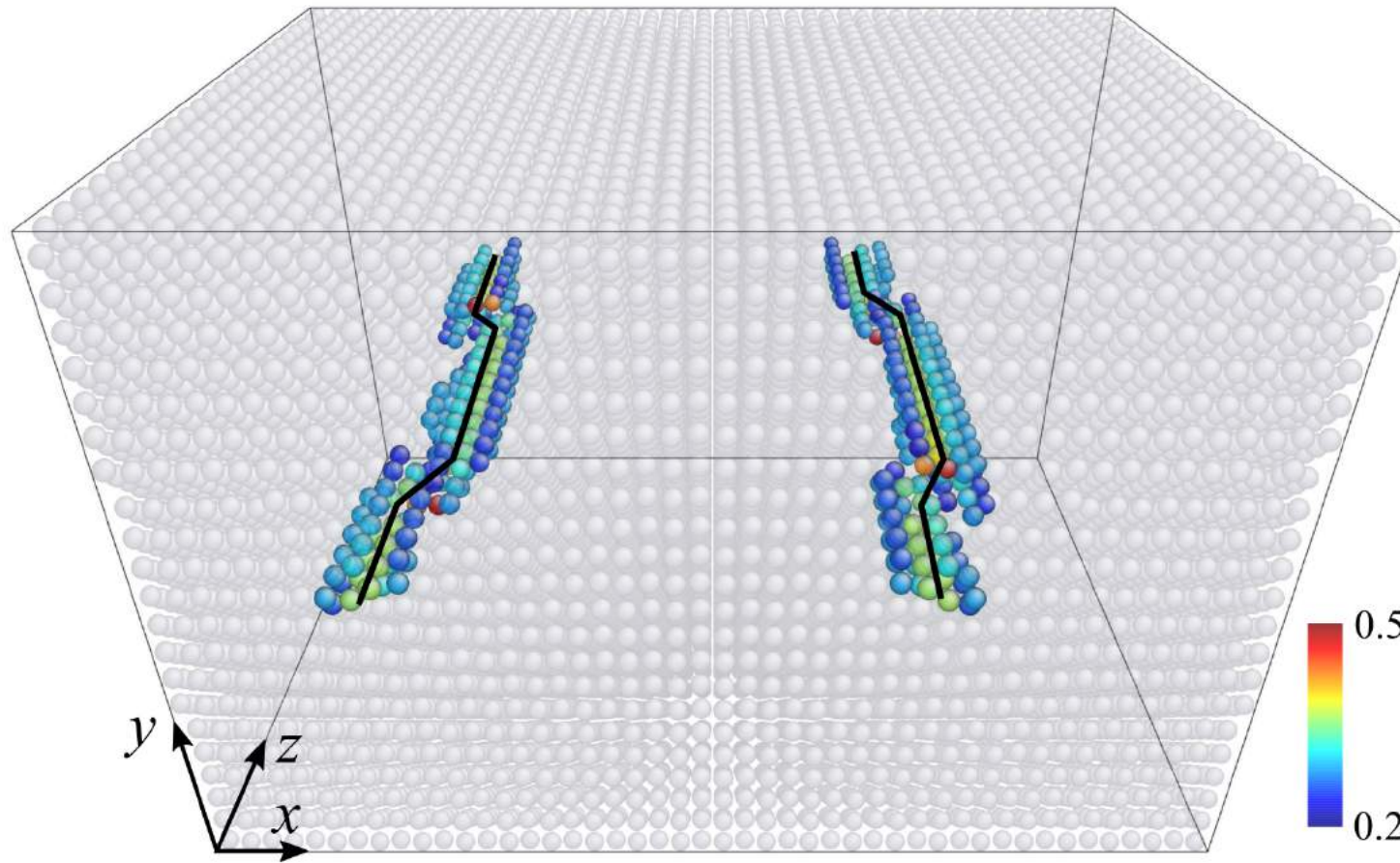


[100](010) in bridgmanite at 30 GPa





# Kink pair nucleation energy computed within periodic cell

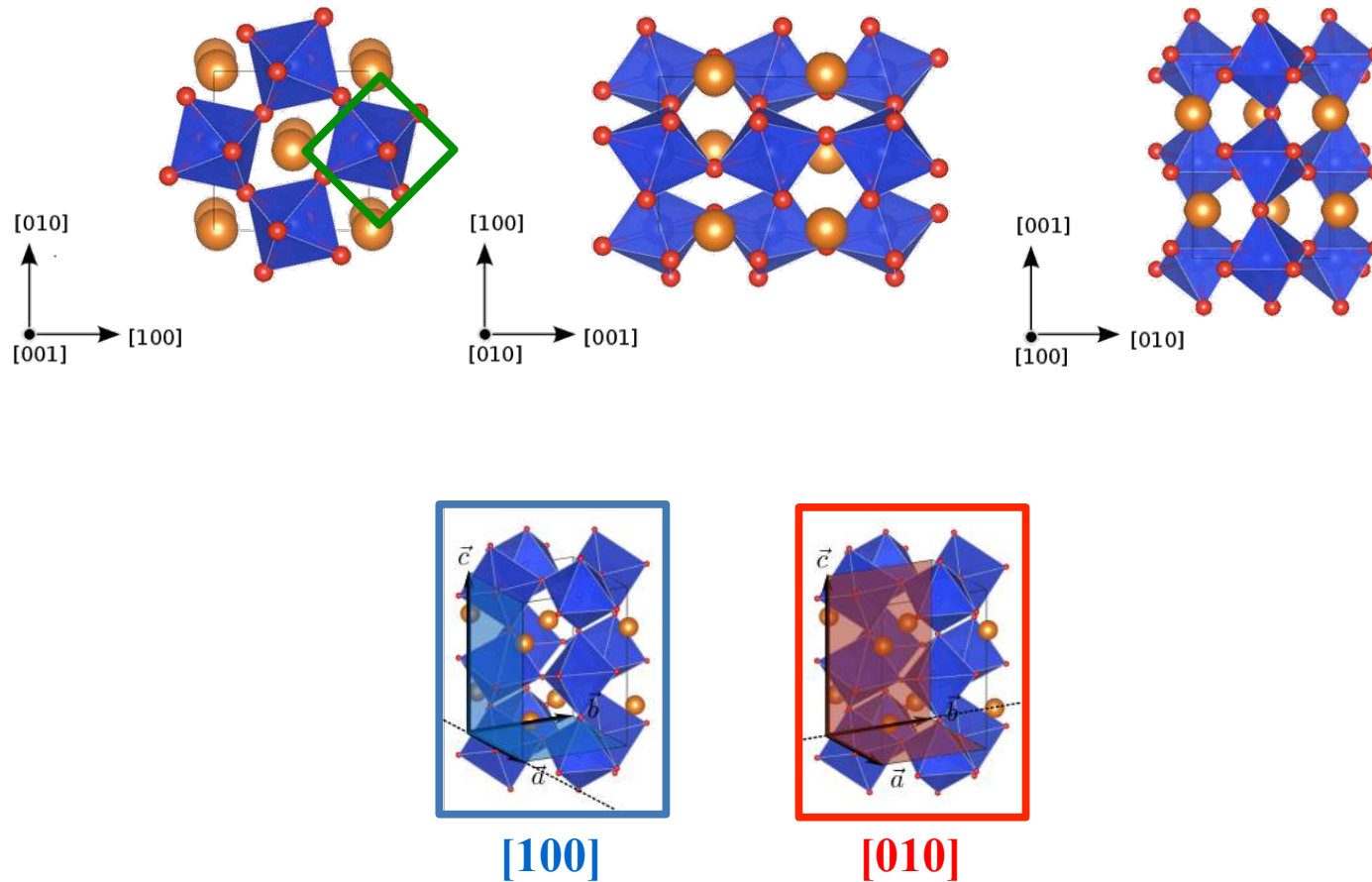


Dipole configuration of kinked dislocation =>  
kink energy  $H_k = 9.5$  eV (P=30 GPa)

$$H_k = \frac{\mu b^3}{2} \sqrt{\tau_P / \mu}$$

*Kraych et al. (PRB 2016)*

# Two slip systems : $[100](010)$ and $[010](100)$

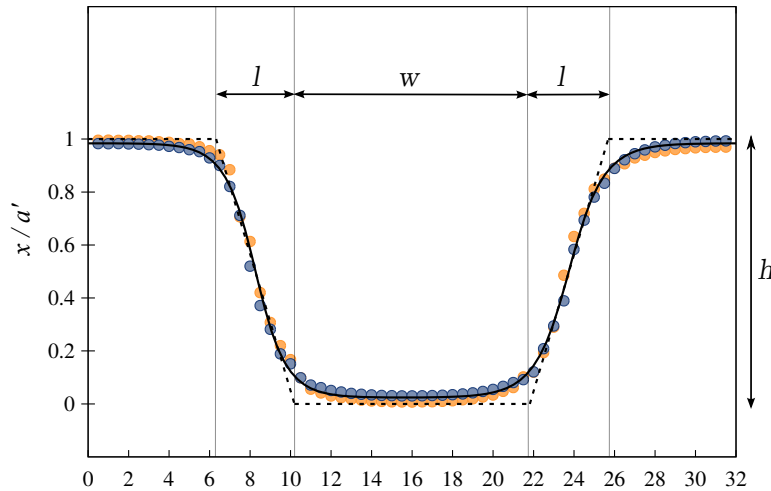


Two slip systems consistent with classical  $\langle 110 \rangle \{ 110 \}$  slip system observed in perfect cubic perovskite (ex.  $\text{SrTiO}_3$ , known to be ductile)



# $\Delta H^*$ function of stress

Kink-pairs configuration: Trapezoidal shape described using  $l$ ,  $w$  and  $h$



$$\begin{aligned}
 2H_k &= \Delta H_k^*(\tau = 0) \\
 &= \frac{\mu b^2}{2\pi} \left[ \frac{l^2 + a'^2/(1-\nu)}{\sqrt{l^2 + a'^2}} \ln \left( \frac{\sqrt{l^2 + a'^2}}{e\rho} \right) \right. \\
 &\quad \left. - l \ln(l/e\rho) + l \ln \left( \frac{2l}{l + \sqrt{l^2 + a'^2}} \right) + l - \sqrt{l^2 + a'^2} \right] \\
 &\quad + 2 \frac{\sqrt{l^2 + a'^2}}{a} \int_0^{a'} V_P(x) dx.
 \end{aligned}$$

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## KINK PAIR NUCLEATION AND CRITICAL SHEAR STRESS

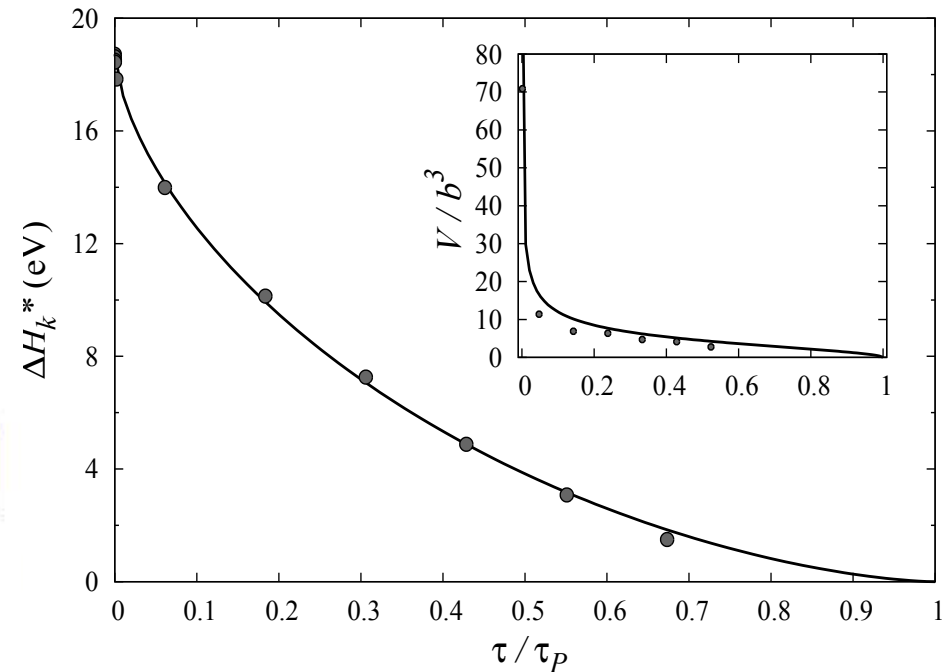
H. KOIZUMI,<sup>1</sup> H. O. K. KIRCHNER<sup>2</sup> and T. SUZUKI<sup>3</sup>

<sup>1</sup>Department of Physics, Meiji University, Higashimita, Tama-ku, Kawasaki 214, Japan,

<sup>2</sup>Institut de Sciences des Materiaux, Université Paris-Sud, F-91405 Orsay, France and

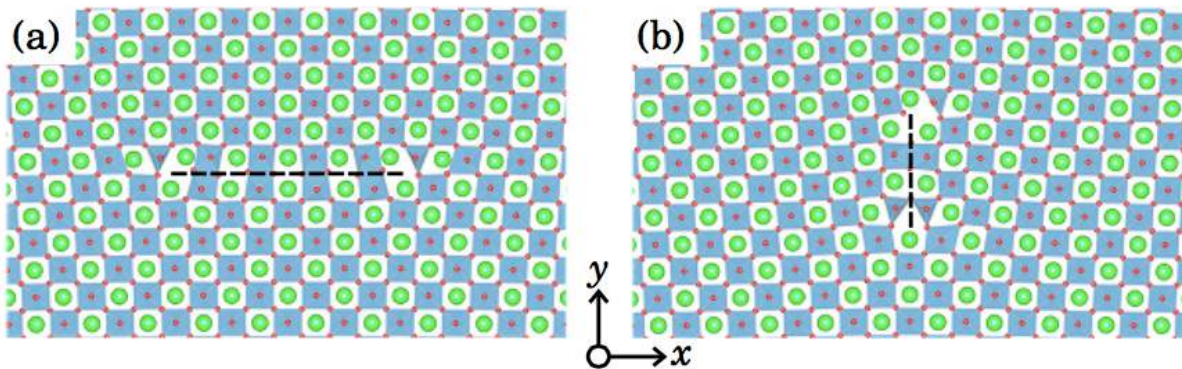
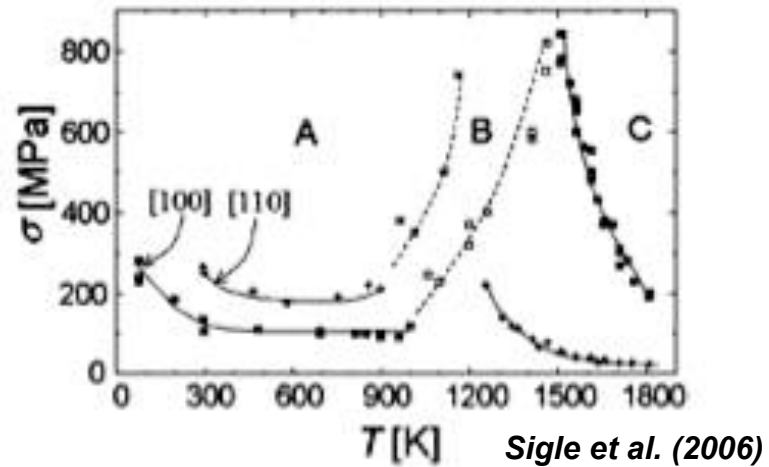
<sup>3</sup>Institute of Industrial Science, University of Tokyo, Roppongi, Minato-ku, Tokyo 106, Japan

$$\Delta H_k = \Delta E_{\text{elas}} + \Delta W_P - W_\tau$$

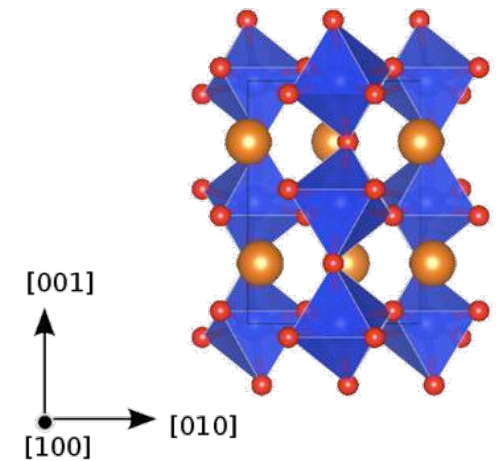
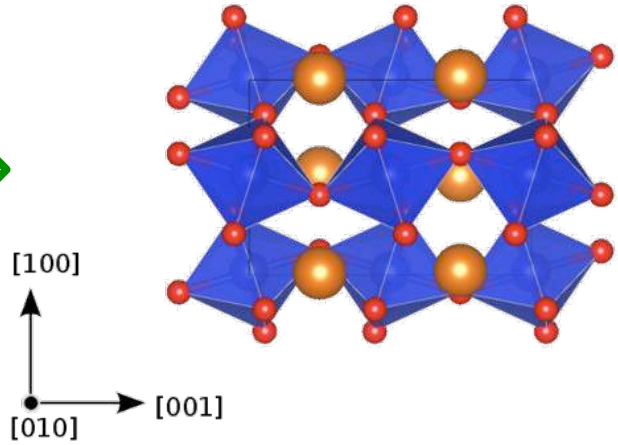
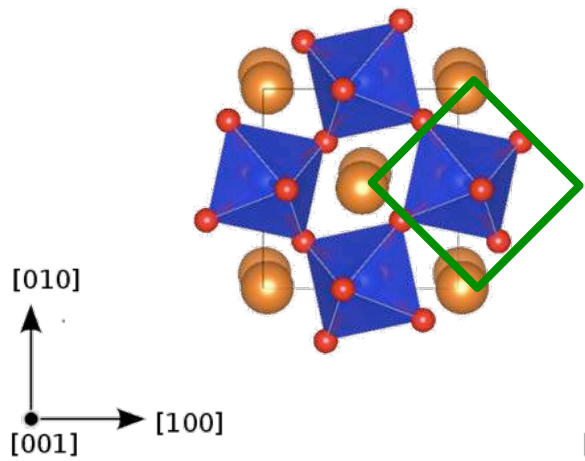


Kraych et al. (PRB 2016)

# SrTiO<sub>3</sub> cubic perovskite mechanical properties

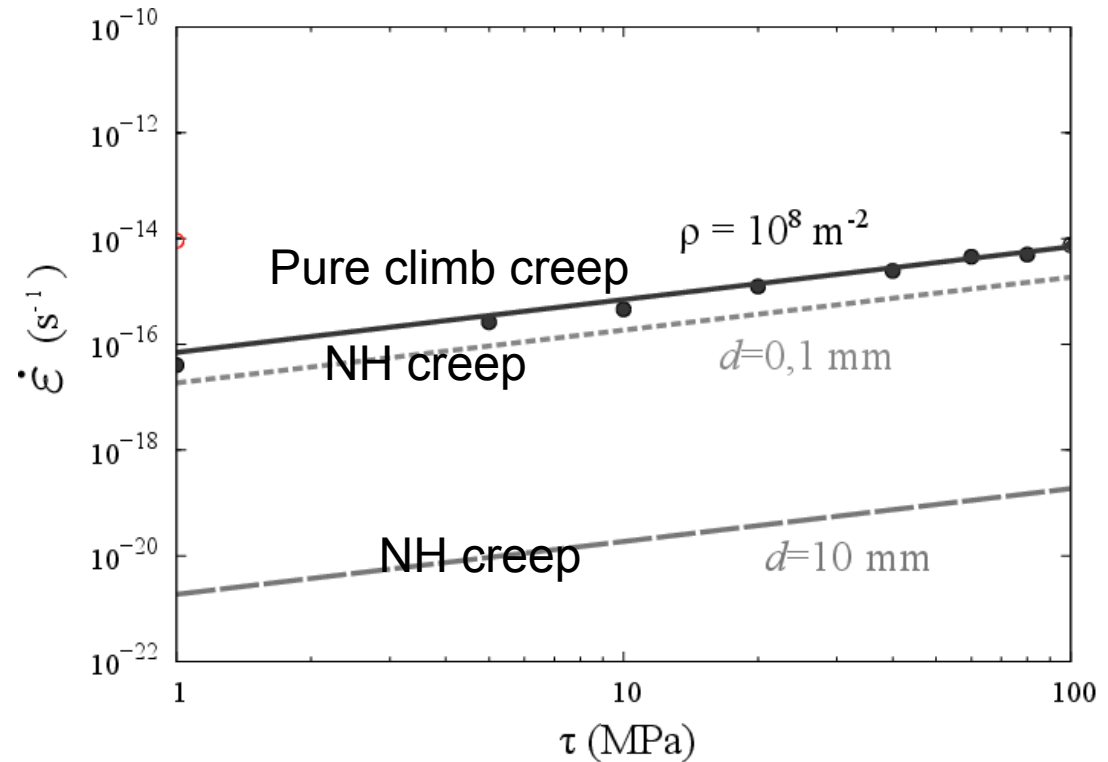
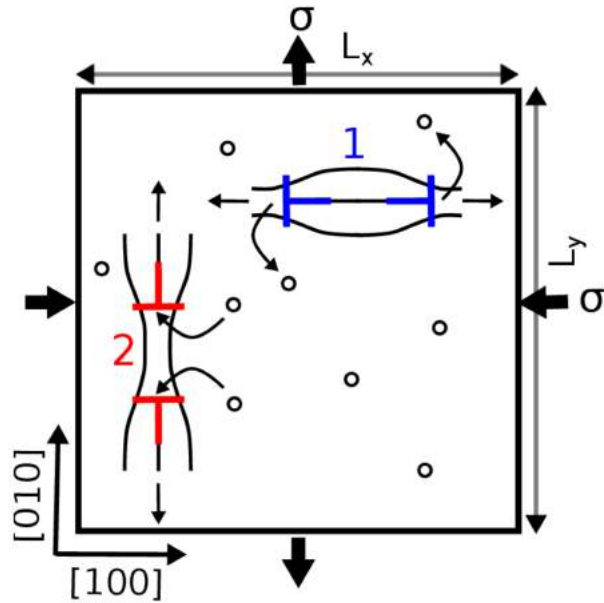


*Hirel et al. (under review Scripta Mat.)*



Cubic		Orthorhombic
$\langle 100 \rangle_c \{010\}_c$	$[100]_c (010)_c$	$[110]_o (\bar{1}10)_o$
	$[010]_c (001)_c$	$[\bar{1}10]_o (001)_o$
	$[001]_c (010)_c$	$[001]_o (\bar{1}10)_o$
$\langle 100 \rangle_c \{011\}_c$	$[001]_c (110)_c$	$[001]_o (010)_o$
	$[001]_c (\bar{1}\bar{1}0)_c$	$[001]_o (100)_o$
$\langle 110 \rangle_c \{001\}_c$	$[110]_c (001)_c$	$[010]_o (001)_o$
	$[\bar{1}\bar{1}0]_c (001)_c$	$[100]_o (001)_o$
$\langle 110 \rangle_c \{1\bar{1}0\}_c$	$[\bar{1}\bar{1}0]_c (110)_c$	$[100]_o (010)_o$
	$[110]_c (\bar{1}\bar{1}0)_c$	$[010]_o (100)_o$

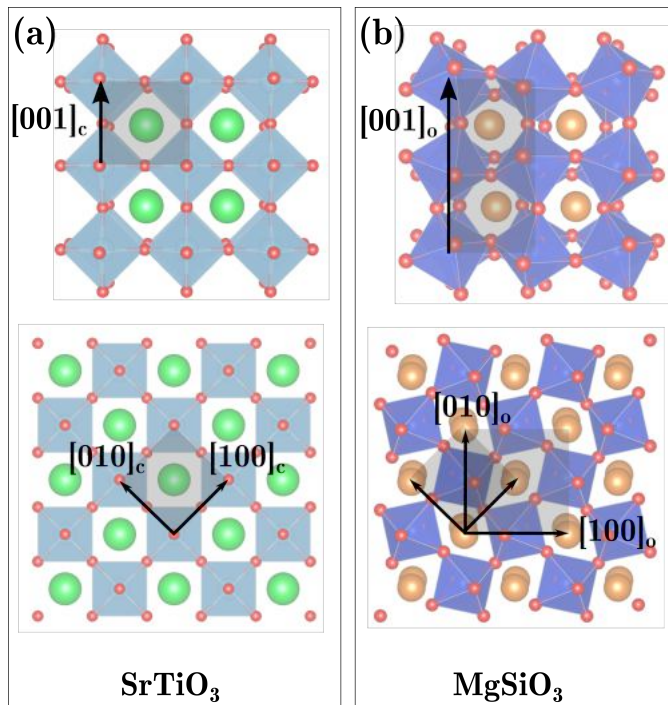
## 2.5 DD simulations of pure climb creep



$d > 0,1$  mm  $\rightarrow$  Pure climb creep  $>$  NH creep

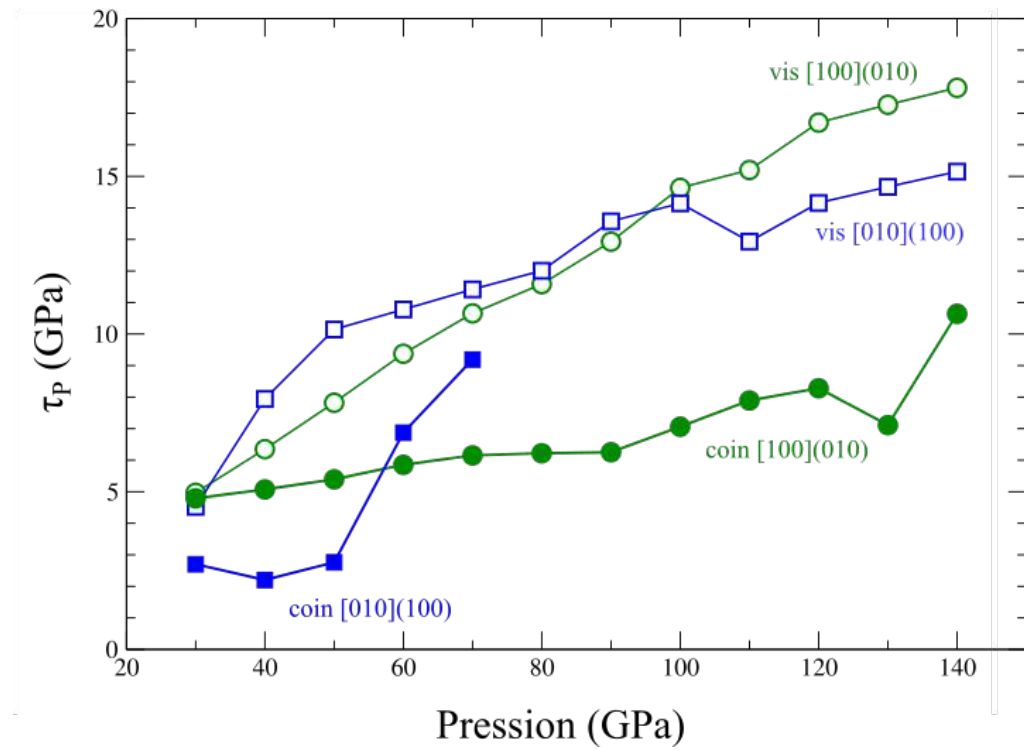
$$\dot{\epsilon} = \alpha \frac{D_{sd} \sigma \Omega}{d^2 k_b T}$$

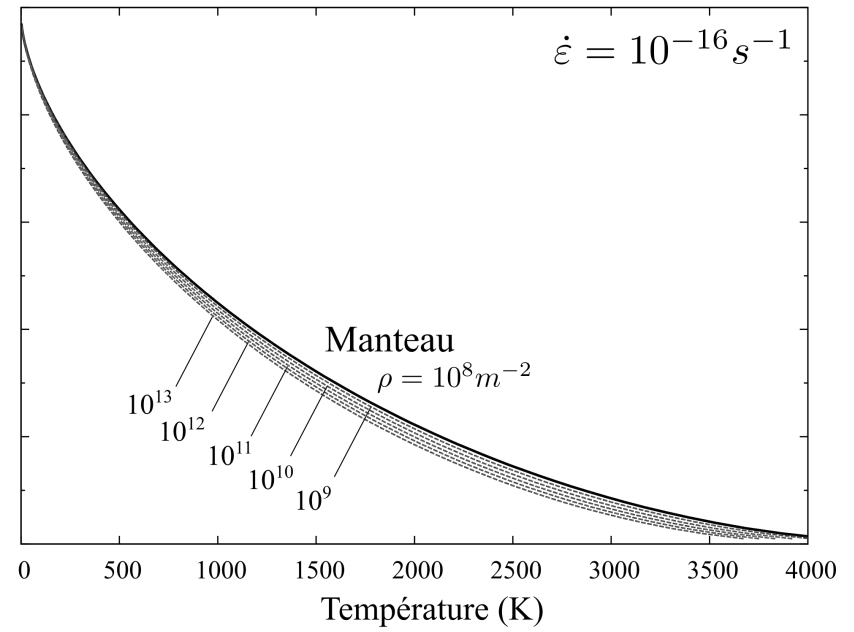
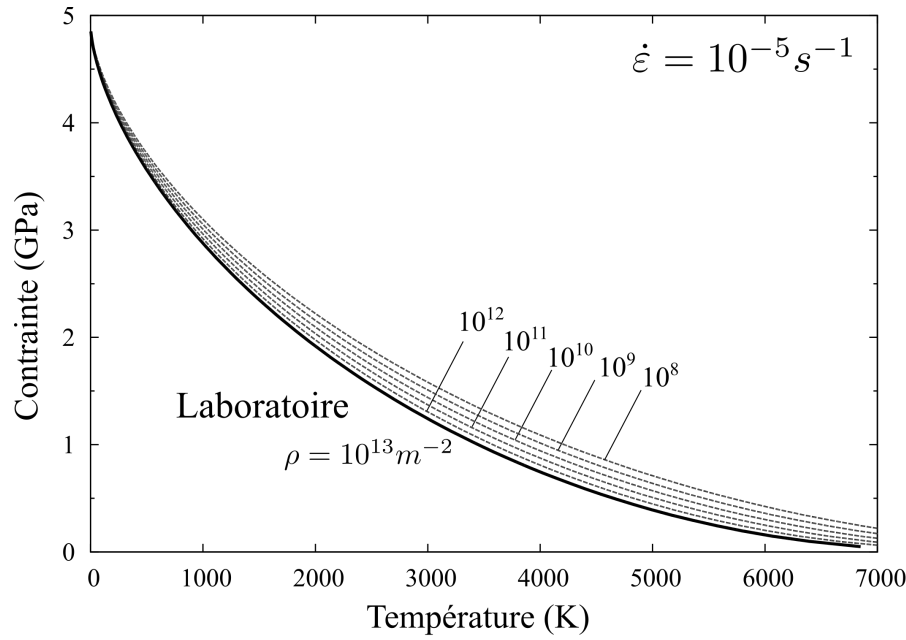
# Slip systems ; orthorhombic -> cubic



$[100]_{\text{ortho}} ; [010]_{\text{ortho}} \rightarrow \langle 110 \rangle_{\text{cubic}}$

$[001]_{\text{ortho}} ; [110]_{\text{ortho}} \rightarrow \langle 100 \rangle_{\text{cubic}}$

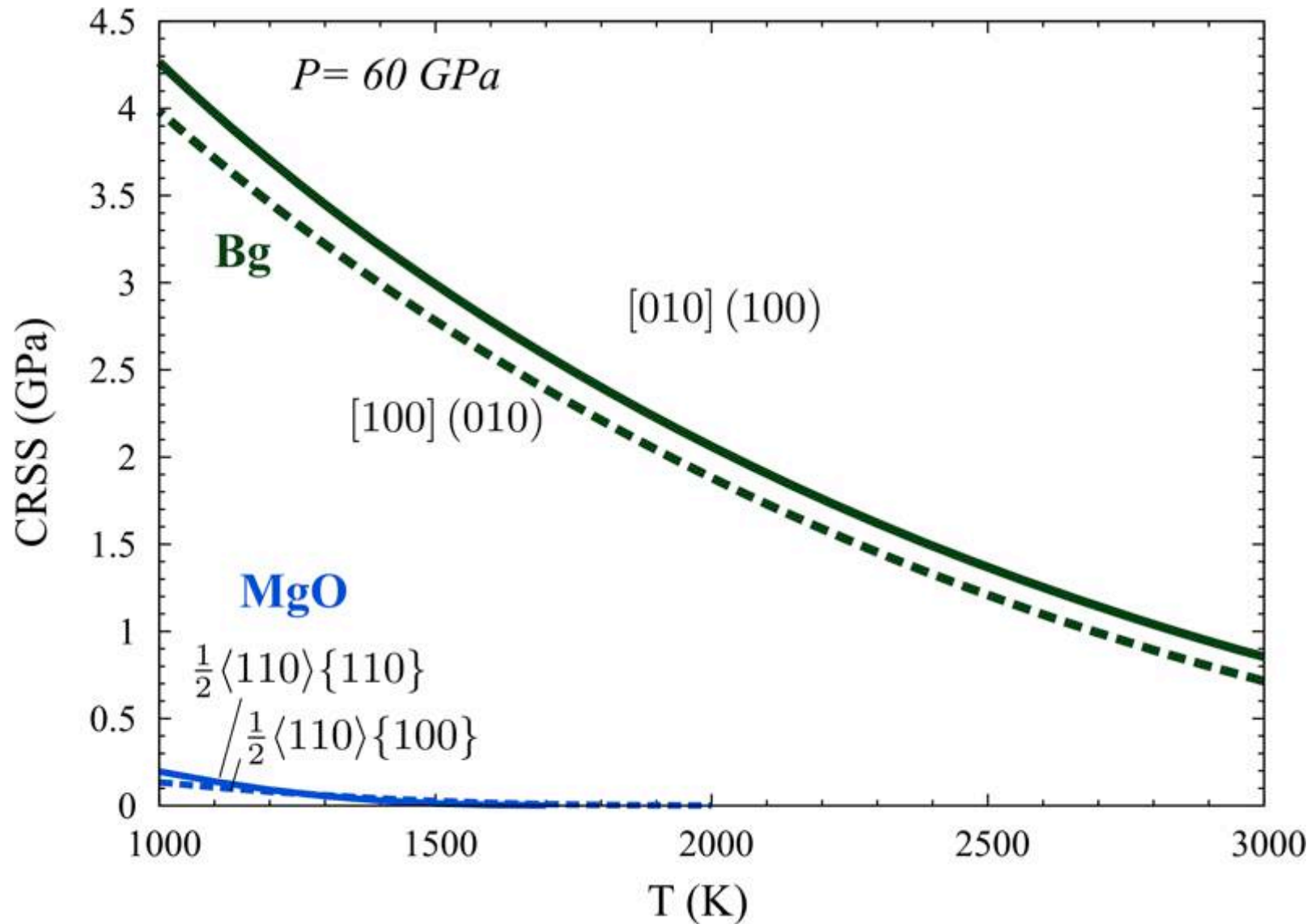




$$\dot{\gamma} = \rho b v_s$$



# MgSiO<sub>3</sub> perovskite (60 GPa)



# MgSiO<sub>3</sub> perovskite (30 GPa)

$$\dot{\epsilon} = 10^{-16} \text{ s}^{-1} ; \rho = 10^8 \text{ m}^{-2}$$

